

Ethanol dehydrogenation to acetaldehyde with mesoporous Cu-SiO₂ catalysts prepared by aerosol-assisted sol-gel

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Abstract:

Copper based catalysts are central for carrying dehydrogenation reactions. However, these materials are prone to deactivation by sintering and coke deposition. Irreversible sintering occurring during reaction (under the effect of temperature) is known to decrease both activity and selectivity, where the unwanted dehydration activity of the support might also play an important role. From this perspective, the quite unreactive silica supports may be attractive. However, using classical catalyst preparation methods (e.g. impregnation), it is a challenge to obtain a stable and homogeneous dispersion of Cu over SiO₂ owing to the weak support-active phase interactions. Taking a sidestep, aerosol-assisted sol-gel is a promising alternative for the facile preparation of mesostructured metallosilicates with high metal dispersion. Here we report, for the first time, Cu-SiO₂ made by the aerosol-assisted sol-gel method and exploited in the ethanol non-oxidative dehydrogenation to acetaldehyde. These catalysts are compared with a series of catalysts made by impregnation to investigate, through a thorough characterization survey, the effect of the synthesis procedure as well as the effect of Cu loading. We show that aerosol-made catalysts do not suffer heavy sintering, reach high ethanol conversions with acetaldehyde selectivity above 75%, and only slowly deactivate upon time due to a (reversible) coking phenomenon.

1. Introduction

Nowadays, many efforts are being pursued to reduce our dependence on fossil resources consumption and to obtain chemicals and fuels from renewable resources. In this field, bioethanol is a prominent platform chemical that can be produced from non-edible biomass feedstocks through fermentation¹. Further upgrading of bioethanol gives access to several important chemical intermediates^{2,3} such as ethylene, acetaldehyde, acetone and hydrogen. Certainly, acetaldehyde is an important target that can then be further transformed to produce valuable fine and bulk chemicals such as carboxylic acids, ketones, and esters⁴. Industrially, it is produced via the well-established Wacker-Hoechst process (oxidizing ethylene to acetaldehyde in the presence of an aqueous solution of palladium chloride and copper chloride). Suitable alternatives – starting from bioethanol – are desirable but stable and robust catalysts for this application are still lacking⁵.

Noble and non-noble metal based catalytic systems have been widely studied for this reaction, and copper appears as the most suitable compromise when considering performances on the first hand and metals availability and price on the second hand⁶. Bueno et al.⁷⁻⁹ reported that the catalytic activity of Cu-ZrO₂ materials depends strongly on Cu nanoparticles size and on their interaction with the support. Hanukovich et al.¹⁰ observe that Cu surface over ZrO₂, TiO₂, AlZrO₂ supports becomes poisoned with reactive intermediates, pointing out the need to choose support with moderate Lewis acidity, to avoid the formation of side products that trigger fouling. Copper-based catalysts supported on Al₂O₃, ZnAl₂O₄, MgAl₂O₄, Zn_xMg_yAl₂O_z have also been investigated¹¹⁻¹⁶. These catalysts are active and selective to acetaldehyde (>90%) also at high conversion, but a progressive deactivation appears inevitable, owing to an interplay between Cu sintering and carbon deposition¹⁷. While sintering is an irreversible effect, mild oxidative treatments could efficiently rejuvenate the catalysts by burning carbonaceous residues. Carbon supported Cu- catalysts¹⁸⁻²⁰ showed good activity levels at 623 K, but dehydration reactions - mainly associated with the presence of oxygen-containing acidic groups on the carbon surface – limit the selectivity of the reaction.

64 Silica-supported Cu-based catalysts are reported as highly selective to acetaldehyde because of the
65 relative inertness of silica towards side reactions such as dehydration ²¹. Thus, Cu/SiO₂ catalysts are
66 also suitable model materials to study the behaviour of Cu nanoparticles ^{9,22–24}. Using such approach,
67 a marked structure-relationship was demonstrated ²⁵. Cu particle size strongly influence selectivity
68 because it dictates the density of (catalytically active) coordinatively unsaturated sites, such as corners
69 and kinks. Nanoparticles around 10 nm have been found to be an optimum for acetaldehyde
70 production while larger particles seem to favour side products such as ethyl acetate ²⁶. Unfortunately,
71 achieving high active phase (Cu) dispersion on silica is complicated with classical synthesis methods
72 and intricate grafting or colloidal catalyst synthesis procedures have to be implemented ^{27–29}.
73 Moreover, these catalysts remain subject to deactivation by coking and are very prone to deactivation
74 by Cu sintering ³⁰. Silica, indeed, as a non-reducible oxide-support, is known to establish only weak
75 interactions with supported metal nanoparticles, and the latter are therefore particularly prone to
76 sintering ³¹. Thus, controlling active phase dispersion and strengthening the interaction with the
77 support is difficult, in particular when applying conventional impregnation methods on pre-formed
78 silica supports. Therefore, the development of new synthetic procedures allowing to (i) control active
79 site speciation and dispersion and (ii) achieve suitable morphological and textural properties ³² is
80 needed.

81 Along this line, the aerosol-assisted sol-gel process (AASG) – which is emerging as a powerful route
82 to produce various nanomaterials and in particular tailored heterogeneous catalysts ^{33,34} – is primed
83 to solve the above-mentioned challenges. The method is based on the atomization and fast reactive
84 drying of a precursor's solution. During processing, inorganic polycondensation reactions take place
85 very rapidly, possibly coupled with the templating action of a sacrificial pore-generating agent. This
86 allows the one-step and continuous production of nanomaterials with desired properties. Aerosol
87 processes have been used to develop catalysts for a wide range of reactions, such as Mo-based
88 catalysts for olefin metathesis ³⁵, TiO₂ catalyst supports for CO₂ methanation ³⁶, Cu-based materials

89 for CO₂ hydrogenation to methanol³⁷, Au based catalysts for CO oxidation³⁸ or RuO₂ based catalysts
90 for electrocatalysis³⁹. While this technique has never been used to synthesize Cu-based catalysts for
91 ethanol non-oxidative dehydrogenation, we surmised that the control offered in terms of Cu
92 dispersion and texture could offer decisive advantages regarding intrinsic activity and stability.
93 Moreover, it must be recalled that aerosol processes are continuous, low-waste, and easily scalable³³.
94 The present work aims at exploiting Aerosol Assisted Sol Gel process (AASG) to prepare low loading
95 Cu-based catalysts for (bio)ethanol dehydrogenation. These new catalysts are compared with
96 conventional impregnated catalysts. With the teachings gained from the deep characterisation of fresh
97 and used catalysts we identify the key parameters that govern the high activity of the AASG-made
98 catalysts, and we identify the causes of deactivation and suitable rejuvenation procedures.

99

100 **2. Experimental section**

101 *2.1 Catalysts preparation*

102 Copper-based catalysts have been synthesized following two different procedures: one-step aerosol-
103 assisted sol-gel (AASG) and incipient wetness impregnation (IWI) of pre-synthesized silica support.

104 **2.1.1 One-pot preparation of Cu-SiO₂ by AASG technique**

105 Solution A is prepared by mixing 0.977 mol of ethanol (VWR, >99.8 v/v%), 0.444 mol of distilled
106 water and 0.0003 mol of Pluronic[®] F127 (Sigma Aldrich, \approx 12600 g/mol)]. A second solution (B) is
107 prepared by mixing 0.057 mol of TEOS (TCI Chemicals, >97.0%), 0.548 mol of an aqueous HCl
108 solution with a pH=2 obtained by dilution of fuming HCl (Roth, 37 wt.%). After preparation, solution
109 A and B were stirred overnight and then mixed together. Cu(NO₃)₂·2.5 H₂O was added in the desired
110 amount to obtain the targeted Cu loading. The obtained solution was then sprayed with an atomizer,
111 supplied by “TSI incorporated[®]”, by applying an air pressure of 207 kPa. The aerosol droplets were
112 dried by passing through a quartz tube heated by a tubular furnace set at 723 K. Then, the dried

113 material was collected on a nitrocellulose filter (Sartorius Stedim, 0.45 μm). Powders were calcined
114 in a muffle furnace under static air, first at 623 K (1 K min^{-1}) for 3 h, and then at 823 K (1 K min^{-1})
115 for an additional 3 h. These catalysts were denoted as A-CuXSi, where X corresponds to the nominal
116 Cu loading on support basis and X = 2, 5, 7.4 and 9.1 wt.%.

117 2.1.2 Cu/SiO₂ catalysts prepared by impregnation

118 Reference catalysts were prepared by incipient impregnation, using a silica support obtained by
119 AASG. The latter was prepared by the same protocol as the CuXSi catalysts but omitting the addition
120 of Cu nitrate. Here, copper is introduced onto the preformed silica support by wetting the latter with
121 an aqueous solution of Cu(NO₃)₂·2.5 H₂O (Alfa Aesar, 98 wt.%). The amount of employed water
122 corresponded to the volume of the support pores, evaluated by N₂ physisorption, incremented by a
123 20% v/v. Powders have been dried and calcined with the same protocol described above. These
124 catalysts were denoted as I-CuXSi, where X corresponds to the nominal Cu loading on support basis
125 and where I stands for “impregnation”.

126 2.2 Materials Characterization

127 Textural properties were measured by N₂ -physisorption at 77 K using a Micrometrics Tristar 3000
128 instrument. Prior to analysis, calcined samples were degassed overnight under vacuum at 443 K. The
129 Brunauer-Emmett-Teller (BET) model was used to determine the Specific Surface Area (SSA, m²/g)
130 in the relative pressure range of 0.05–0.30. Total pore volume (V_p) was estimated from the adsorption
131 branch of the isotherm at p/p₀=0.98 and the average pore diameter (D_p) was estimated from the BJH
132 model applied on the adsorption isotherms.

133 XRD analysis was carried out by means of a Bruker D8 Advance diffractometer (Bragg–Brentano
134 geometry) on both fresh and spent catalysts. The diffractometer works with a Cu K α source
135 (λ =0.15418 nm) at 1200 W (30 mA, 40 kV). Diffraction patterns have been acquired by setting the
136 following parameters: 2 θ range 5°–100°, step size 0.05° (2 θ) and 1.5 s each step. The detector was a

137 Bruker Lynxeye XE-T. Identification of the phases was carried out using Pearson's Crystal Database
138 ⁴⁰. Crystallites sizes of CuO and of Cu have been evaluated on the most intense peak by Scherrer
139 formula, via DIFFRAC.EVA V4.2.1 software. For CuO, peak at $2\theta = 35.66$ has been considered, for
140 Cu, peak at $2\theta = 43.40$ has been considered.

141 ICP-AES analyses were performed on an ICP Thermo Scientific 6500 instrument after dissolution of
142 samples (≈ 100 mg) by metaborate Li - tetraborate Li fusion.

143 Scanning Electron Microscope Zeiss SUPRA 40 VP, equipped with a field emission gun (FE-SEM)
144 was used to investigate sample morphology and composition for both fresh and spent catalysts. This
145 microscope is equipped with a high sensitivity "InLens" secondary electron detector and with a EDX
146 (Energy Dispersive X-Ray) Spectrometer OXFORD "INCA Energie 450 \times 3". Samples were
147 suspended in ethanol. A drop of the resultant mixture was deposited on a Lacey Carbon copper grid
148 and the dried sample was then imaged.

149 Samples composition was investigated by using a scanning electron microscope (SEM) Zeiss Evo 40
150 equipped with a Pentafet Link Energy Dispersive X-ray Spectroscopy (EDXS) system managed by
151 the INCA Energy software (Oxford Instruments, Analytical Ltd., Bucks, U.K.). Samples powders
152 were directly mounted on a high purity conductive double sided adhesive carbon tab.

153 TEM and STEM-EDS measurements were carried out on a FEI Titan Themis instrument with a
154 combination of a spherical aberration image (Cs) corrector, a monochromator system, sensitive
155 ChemiSTEM technology, and a high-end GATAN GIF Quantum Energy Filter for EELS and EFTEM
156 with a new enhanced piezo stage, FEI and GATAN software, and a FEI Ceta 16-megapixel CMOS
157 camera. Powders have been dispersed in cyclohexane and 4 μ L deposited on a gold grid covered by
158 QuantiFoil[®] holey carbon and then dried in air.

159 X-ray photoelectron spectroscopy (XPS) analyses were carried out at room temperature with an SSI-
160 X-probe (SSX 100/206) photoelectron spectrometer from Surface Science Instruments (USA),

161 equipped with a monochromatic Al K X-ray microfocused source (1486.6 eV). Samples were deposited
162 onto small sample holders with adhesive tape and then placed on an insulating ceramic carousel
163 (Macor[®], Switzerland). Charge effects were avoided by placing a nickel grid above the samples and
164 using a flood gun set at 8 eV. The binding energy scale was calibrated by fixing the Si 2p peak at
165 103.5 eV⁴¹. Data treatment was performed using the CasaXPS program (Casa Software Ltd., UK).
166 The peaks were decomposed into a sum of Gaussian/Lorentzian (85/15) after subtraction of a Shirley-
167 type baseline.

168 Infrared (IR) spectra have been obtained with a Nicolet 380 Fourier Transform IR spectrometer.
169 Sample powders were pressed into thin wafers with KBr and spectra were recorded in air (0.5 wt%
170 sample in KBr). Spectra have been collected with 100 scans with a resolution of 2 cm⁻¹.

171 DR-UV-vis-NIR spectra were collected with a JASCO V570 instrument equipped with an integrating
172 sphere. The powders were gently pressed in the sample holder.

173 H₂ Temperature Programmed Reduction (TPR) measures were conducted using a CATLAB
174 instrument, from Hiden equipped with QGA mass spectrometer for gas analysis. 55 mg of sample
175 were loaded in a quartz reactor. First the sample was maintained at 373 K for 1 hour under flowing
176 argon (40 mL/min). Next, the samples were exposed to a 1% H₂ flow in argon (40 mL/min) and then
177 heated up to 873 K using a heating rate of 5 K/min. The H₂ uptake is calculated by integration of the
178 H₂ consumption profiles. Signal intensities were referenced to the calibrated signals for known
179 volumes of analyzed gases.

180 NH₃ Temperature programmed Desorption (TPD) measurements were conducted in the same
181 CATLAB instrument. 40 mg of powders were loaded in a quartz reactor. A reductive pretreatment
182 was carried out for 30 min at 573 K, under a 30 mL/min 5 vol. % H₂ in argon. For silica and the
183 zeolite catalyst employed as acid benchmark, the hydrogen pretreatment was not applied. Next, the
184 samples were cooled to 333 K under inert atmosphere. Then they are exposed to a 2 vol. % NH₃ in
185 argon (30 mL/min) flow for 1 hour, and then flushed under argon (30 mL/min) at 333 K for 60 min.

186 The temperature-programmed desorption was carried out up to 1023 K with a ramping rate of 10 K
187 /min in a 30 mL/min flow of Ar. The NH₃ quantification was made by integration of the desorption
188 profile. Signal intensities were referenced to the calibrated signals for known volumes of analyzed
189 gases.

190 Spent catalysts were studied by thermo-gravimetry, using a TGA/DSC 3+ apparatus from Mettler-
191 Toledo, coupled with a Pfeiffer Vacuum ThermoStarTM mass spectrometer. Samples (\approx 20 mg) were
192 placed in alumina crucibles (70 μ L) and heated at 10 K min⁻¹ from room temperature to 1173 K under
193 a dry air flow (100 mL/min). The following m/z have been recorded over the whole experiment: 28
194 (N₂), 18 (H₂O), 32 (O₂) and 44 (CO₂).

195 *2.3 Ethanol dehydrogenation to acetaldehyde*

196 Catalytic tests were carried out in a fixed bed reactor (stainless steel, 0.6 cm internal diameter). 50
197 mg of calcined catalysts (pressed, crushed and sieved in the 0.20–0.40 mm range) were diluted with
198 glass beads (0.5–1.0 mm), by keeping a constant catalytic bed volume. Before reaction, catalysts were
199 pre-reduced in situ by feeding hydrogen (20 vol.% H₂ in N₂, total flow rate of 175 mL/min) for 30
200 min at 573 K with a ramping rate of 5 K min⁻¹. Then, the reactor was set at the desired reaction
201 temperature (473 K) and the reaction feed was introduced: absolute ethanol was fed through a syringe
202 pump (NE-300) in a stream on N₂, achieving the desired molar fraction in the gas phase (2.5% mol,
203 total flow rate of 120 mL/min). All measures were carried out at atmospheric pressure while
204 temperatures were increased in a stepwise manner, from 473 K to 673 K. Every temperature step was
205 kept for 1 hour, ensuring steady state conditions. Effluent gas was analysed online with a VARIAN
206 3800 Gas Chromatograph (Varian Medical Systems, Machelen, Belgium). Four injections were made
207 at each temperature. The injector was set at 423 K and the GC was equipped with a flame ionization
208 detector (FID) kept at 463 K, a Restek Rt-U-BOND column (30 m long, internal diameter of 0.32
209 mm, film thickness of 10 μ m) and a programmable oven (setup: 383 K for 5 minutes, then to 423 K
210 with a ramping rate of 5 K/min, dwelled for 5 minutes).

211 Stability tests on A-Cu_{7.4}Si and I-Cu_{7.4}Si have been carried out at 573 K for 8 hours in the same
212 conditions. In the case of A-Cu_{7.4}Si catalysts, only 40 mg of catalyst was introduced in order to have
213 iso-conversion comparison with the impregnated catalyst (I-Cu_{7.4}Si)

214 Ethanol conversion (X_{EtOH}) is defined as:

215
$$X_{\text{EtOH}} = (n_{\text{EtOH(in)}} - n_{\text{EtOH(out)}}) / n_{\text{EtOH(in)}} \quad (1)$$

216 selectivity to product i is defined as:

217
$$S_i = n_i / (v_i (n_{\text{EtOH(in)}} - n_{\text{EtOH(out)}})) \quad (2)$$

218 And yields are defined as:

219
$$Y_i = n_i / v_i n_{\text{EtOH(in)}} \quad (3)$$

220 where n_i is the molar flow of compound i, and v_i is the ratio of stoichiometric reaction coefficients.

221 The error in the steady state carbon balance is $\pm 5\%$.

222 *2.4 Regeneration test*

223 After the stability test (8 h at 573 K) an oxidation step was carried out by flowing a gaseous stream
224 (50 mL/min) of 20 vol % O₂ in N₂. Temperature was raised with a ramping rate of 5 K min⁻¹ up to
225 788 K, and then dwelled for 15 minutes. After cooling down up to room temperature, the catalyst has
226 been tested again in the same experimental conditions.

227

228 **3. Results and discussion**

229 *3.1 Catalysts characterisation*

230 The copper content has been experimentally verified by ICP-AES (Table 1). Experimental data are
231 in good agreement, even if the observed loading is slightly lower than the nominal one (for both

aerosol-made catalysts and impregnated catalysts), suggesting a contribution of the hygroscopic properties of the metal precursor used in the preparation.

Nitrogen physisorption curves (Figure 1A, C) revealed that all the calcined catalysts showed Type IV isotherms indicating the presence of open mesopores. The shape of the hysteresis loop corresponds to the H2-type (asymmetrical with a steeper desorption branch than the adsorption one) characteristic of restrictions within the porous network. The forced closure of the hysteresis loop at p/p_0 values of 0.4–0.5 indicated the presence of mesopores smaller than ~ 4 nm⁴². The Barrett-Joyner-Halenda (BJH) model applied on the adsorption branch (Fig. 1B, D) of the isotherm revealed the presence of mesopores in the 6-12 nm range (centred at about 8 nm) for all catalysts. The nitrogen uptake at low relative pressure also indicates the presence of micropores whose contribution remains relatively small, as underlined by textural values summarised in Table 1.

Table 1: Composition and textural properties of catalysts prepared via IWI (I) and in one step via the AASG process (A).

Sample	Cu wt.% - ICP-AES	Surf. area [m ² ·g ⁻¹]	V _p ^[a] [cm ³ ·g ⁻¹]	V _{micro} ^[b] [cm ³ g ⁻¹]	D _p ^[c] [nm]
A-SiO ₂	/	380	0.45	0.06	7
I-Cu ₂ Si	1.3	329	0.44	0.05	7
I-Cu ₅ Si	3.9	300	0.37	0.03	7
I-Cu _{7.4} Si	6.4	290	0.54	0.03	11
I-Cu _{9.1} Si	8.0	297	0.40	0.03	7
A-Cu ₂ Si	1.6	505	0.70	0.09	9
A-Cu ₅ Si	3.6	458	0.42	0.11	6
A-Cu _{7.4} Si	6.7	458	0.62	0.10	8
A-Cu _{9.1} Si	8.2	418	0.66	0.09	9

247 [a] Total pore volume, estimated from the adsorption branch of the isotherm at $p/p_0=0.98$. [b] Microporous volume,
248 estimated from the t-plot. [c] Average pore diameter, estimated from the BJH model applied on the adsorption isotherms.

249

250 The specific surface area (SSA) and pore volume were generally higher for A-catalysts (prepared via
251 the one-pot AASG process), reaching as high as $505 \text{ m}^2/\text{g}$ for A-Cu₂Si. Interestingly, the micropore
252 volume appears to be more important for one pot catalysts than for bare silica or impregnated
253 materials. A possible explanation is related to the direct introduction of copper nitrate in the formation
254 of A-CuX catalysts, trapped in the porous structure during the rapid droplet's condensation and drying
255 step. Since Cu²⁺ ionic radius is -larger than that of Si⁴⁺, it is possible that the silica network rearranges
256 to host copper species, creating roughness at the surface. Alternatively, nitrates may behave as a
257 "foaming agent" during calcination creating a family of smaller pores upon the departure of NO_x
258 species. Upon increase in copper loading, specific surface area and pore volume tend to decrease,
259 which is logical, considering the fact that the fraction of low-SSA Cu NP increases at the expense of
260 the fraction of high-SSA silica.

261

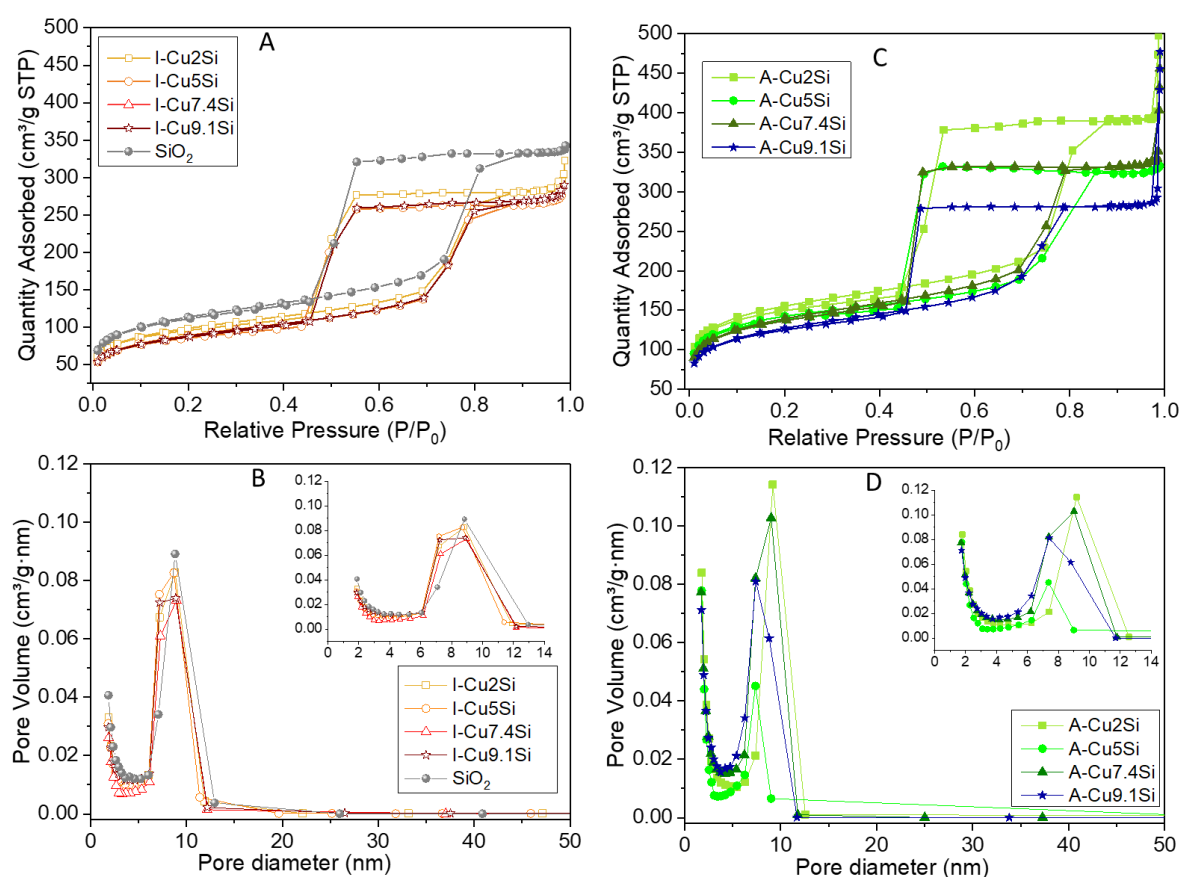
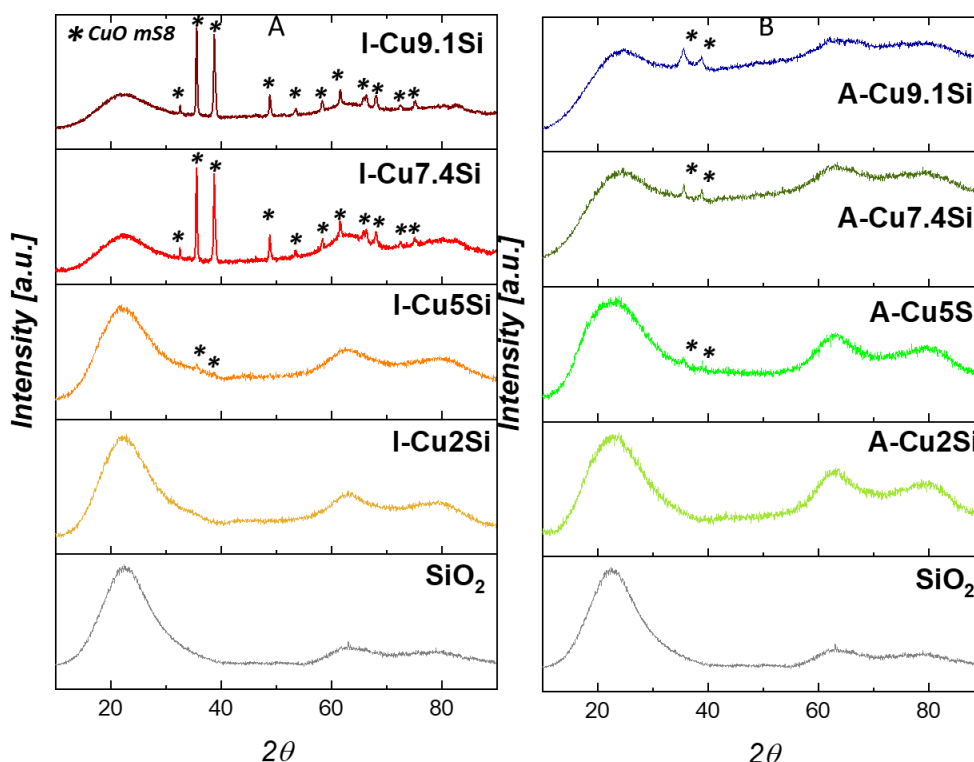


Figure 1: N₂ adsorption/desorption isotherms (insets A and C) and pore size distributions (B and D) of investigated catalysts.

The diffraction patterns were recorded for all fresh catalysts (Figure 2A and 2B). For both I-CuXSi and A-CuXSi series of catalysts, a broad diffraction peak at about $2\theta = 22^\circ$ is attributed to the amorphous silica support⁴³. In all catalysts having a Cu content equal or higher than 5 wt.%, the CuO phase (tenorite, mS8; PCD ref. 1243879)⁴⁰, is observed. Expectedly, the CuO diffraction peaks intensity increases with the increasing copper content in the catalyst. The Scherrer equation, applied on the most intense CuO peak at $2\theta = 35.6^\circ$, indicates an average crystallite size of 32-33 nm for both I-Cu7.4Si and I-Cu9.1Si samples. In the XRD patterns of samples obtained by AASG synthetic procedure (Fig. 2 B), the peaks intensity of CuO phase is lower; in fact, only the two main peaks, at $2\theta=35.8^\circ$ and 38.8° , are visible and broader than those observed for I-CuXSi series. For A-Cu7.4Si

275 and A-Cu9.1Si, the crystal size evaluation with Scherrer equation leads to an average size of about
 276 10 nm. For low loading catalysts, it is only possible to hypothesize the presence of nanoparticles with
 277 an average size equal or lower than 10 nm, hypothesis that is consistent with TEM images (see below
 278 and ESI).



279
 280 **Figure 2:** XRD patterns of the fresh I-samples (inset A), and A-samples (inset B). Please note that broad and weak
 281 peaks around $2\theta = 63^\circ$ and 79° are attributable to the fixing agent used in the sample preparation for the analysis.

282
 283 FE-SEM images of bare silica support shows spherical silica particles, with diameters between 0.1 –
 284 5 μm (Figure 1S(A)). The obtained spherical shape is a reminiscence of the synthetic procedure that
 285 relies on a drying step of spherical aerosol droplets. In low loading impregnated catalysts i.e., I-Cu2Si
 286 and I-Cu5Si, no bright crystalline aggregates have been detected (Figure 1S(B) and (C)). At higher
 287 loading i.e., I-Cu7.4Si and I-Cu9.1Si, bright crystals are clearly distinguishable and might be ascribed
 288 to CuO (Fig. 3). Indeed, STEM energy dispersive X-ray spectroscopy (STEM-EDXS) confirms the

289 presence of Cu-rich aggregates that are correlated with well-visible crystallites. These particles come
 290 out from the silica matrix, often sandwiches between spherical silica particles (Fig. 3A and 3C). The
 291 morphology of the latter remains unaffected upon impregnation (Fig. 1S). In fact, focusing on the
 292 smooth silica microspheres (away from the Cu aggregates), EDX spectroscopy reveals a fixed copper
 293 amount (Fig. S1(D)), equal to ≈ 3.0 wt.%, for I-Cu5Si, I-Cu7.4Si, and I-Cu9.1Si. This agrees with
 294 XRD results, where the intensity of CuO peaks is very high for I-Cu7.4Si and I-Cu9.1Si specimens,
 295 due to the presence of isolated CuO crystals outside of the silica spheres. To sum up, starting from I-
 296 Cu5Si, only a fraction of the copper is introduced into the silica microspheres by means of the
 297 impregnation step and a significant fraction accumulates outside of the support particles, in the form
 298 of CuO crystals.

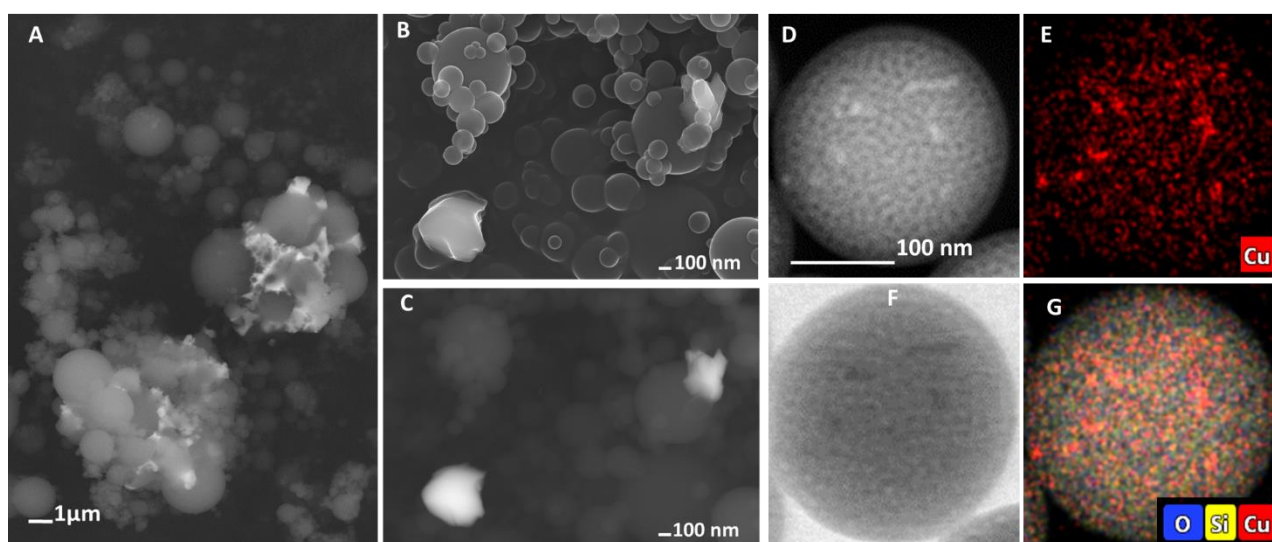
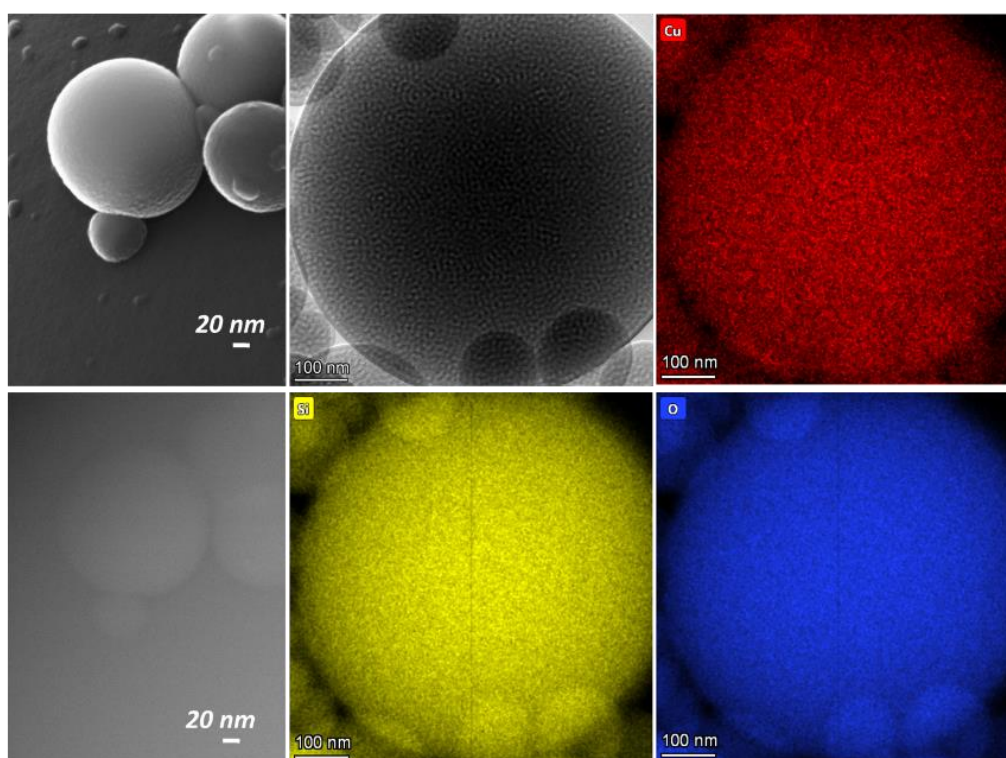


Figure 3: FE-SEM micrographs obtained on I-Cu7.4Si (A, C using the BSE signals, and B using the SE signal) and
 STEM images obtained on I-Cu5Si (D in dark field, E = Cu mapping, F in bright field, G = O, Si, Cu mapping).

For aerosol-made catalysts (see the example of A-Cu7.4Si in Fig. 4 and other images for A-Cu5Si in
 ESI, Fig. S2), FE-SEM microphotographs do not reveal the presence of copper-containing aggregates.
 This is in full agreement with XRD analyses. Moreover, by looking at EDX analysis for A-catalysts,
 the copper amount into the spheres agrees with the total Cu amount in the specimen (Fig. S2, F-G).

307 Thus, in agreement with FE-SEM microphotographs, it is confirmed that all the copper is
 308 incorporated into the silica microspheres. Elemental mapping by means of STEM-EDXS
 309 unambiguously confirms the homogeneous distribution of copper throughout the silica microspheres
 310 (Fig. 4).



311
 312 **Figure 4:** FE-SEM microphotographs of A-Cu_{7.4}Si using SE (left, top) and BSE (left, bottom), STEM images of A-Cu_{7.4}Si
 313 with elemental mapping (right).

314

315 Insight on surface copper chemical state has been obtained by X-ray photoelectron spectroscopy
 316 (XPS) (Table S1 and Fig. 5). In agreement with the literature⁴⁴, Cu 2p_{3/2}, Cu 2p_{1/2} peaks are found at
 317 ~933.3 eV and ~952.5 eV, accompanied by the shake-up satellite peaks (~943 eV and ~963 eV),
 318 indicating the presence of CuO (Cu²⁺ with a d⁹ electron configuration), in agreement with the adopted
 319 synthetic route (calcination in air) and with XRD results. As already reported⁴⁵, two contributions
 320 can be seen for Cu 2p_{3/2}, indicating the existence of two Cu (II) species with different chemical
 321 environment at the catalyst surface. The first one at ~933 eV is typically due to bulk-like CuO species

322 while the second feature at higher binding energy (~ 936 eV) is indicative of well dispersed Cu (II)
 323 interacting with the silica network. This may suggest the presence of $-\text{Cu}-\text{O}-\text{Si}-$ species (i.e. well
 324 dispersed CuO and or copper phyllosilicate-like species). While this feature at 936 eV is also seen in
 325 impregnated catalysts, as shown in table 1S its intensity is much smaller than in aerosol-made
 326 catalysts, pointing again to a higher dispersion of copper via the AASG synthesis procedure. In fact,
 327 for A-catalysts upon increasing of the loading we linearly increase the surface Cu concentration (ESI,
 328 Fig. S3) while over impregnated catalysts there is no increment upon Cu loading. According to
 329 microscopy characterization and XRD patterns, this means that via impregnation only a certain
 330 amount of Cu is introduced into the pores of the silica microspheres and well dispersed at the surface.
 331 After that, Cu start to heavily aggregate without further increasing Cu superficial concentration.

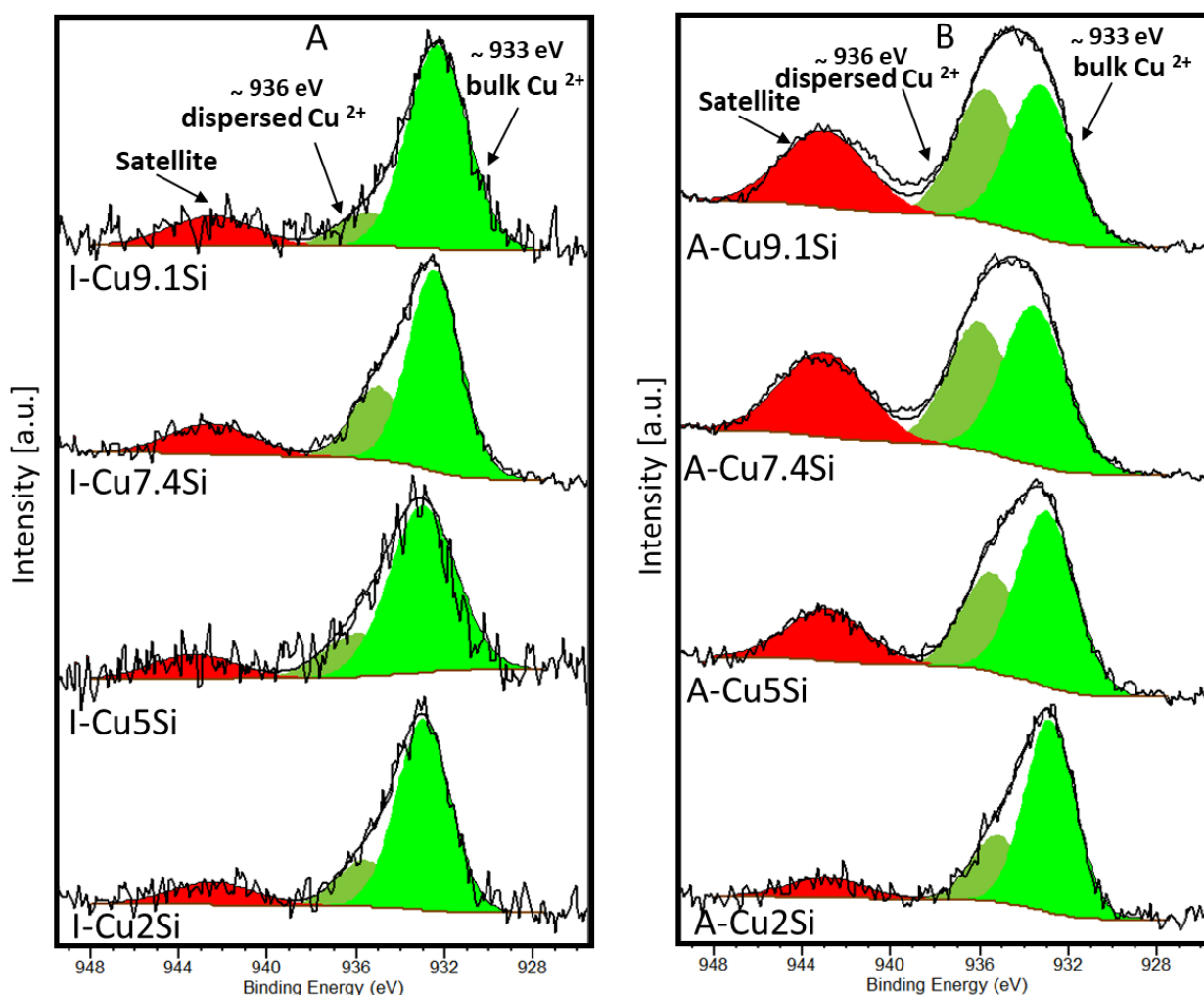


Figure 5: High-resolution XPS spectra of Cu $2p_{3/2}$ species on fresh I-catalysts (A) and A-catalysts (B).

334 H₂-TPR measurements were performed on A-Cu_{9.1}Si and I-Cu_{9.1}Si to probe copper accessibility
335 and reducibility (Fig. S4). The hydrogen consumption for I-Cu_{9.1}Si was 1.37 mmol H₂/g_{cat}, matching
336 closely the amount of Cu present in the sample (1.26 mmol Cu/g_{cat}) and indicating the accessibility
337 of Cu. For A-Cu_{9.1}Si, 1.07 mmol H₂/g_{cat} of hydrogen was consumed, slightly lower than the actual
338 Cu loading is 1.30. Thus, most of the copper species are reduced, even if it is likely that a small
339 fraction of copper species remains inaccessible (trapped in the silica matrix). At the same time, copper
340 reduction occurs at lower temperature in A-Cu_{9.1}Si, as compared to I-Cu_{9.1}Si. This easier reduction
341 suggests that CuO nanoparticles are smaller in the A-catalyst.⁴⁶

342 FT-IR spectra featured the characteristic features of amorphous silica absorption bands for both series
343 of catalysts. The IR spectrum of silica is characterized by the bands assigned to the Si-O-Si
344 asymmetric stretching ($\nu_1 = 1091\text{ cm}^{-1}$ with the corresponding pronounced shoulder at 1230 cm^{-1}),
345 the Si-O stretching of SiOH group (968 cm^{-1}), the coupled Si-O-Si symmetric stretching/in plane
346 bending ($\nu_2 = 803\text{ cm}^{-1}$), and ~~the~~ corresponding out of plane bending (rocking) mode ($\nu_3 = 467\text{ cm}^{-1}$)
347 ¹⁾ ⁴⁷. The spectrum obtained for the bare aerosol-made silica support is almost unchanged after
348 impregnation of copper, showing that in this case the bulk silica is not significantly perturbed. In
349 contrast, in the case of A-Cu_XSi catalysts, the position of the ν_1 mode is shifted to 1085 cm^{-1} ,
350 indicating that in this case the bulk is perturbed, probably because of the formation of Cu species in
351 strong interaction with the silica network^{48,49}.

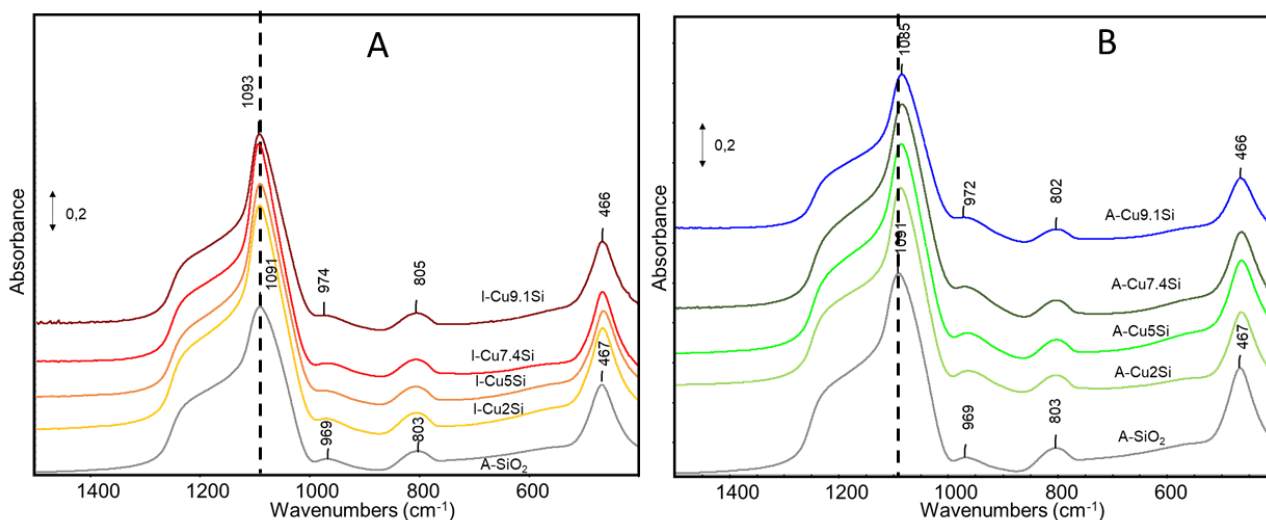


Figure 6: FTIR skeletal spectra (KBr pressed disks) of the fresh I- (A) and A- (B) catalysts.

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355 UV-vis spectroscopy (Fig. S5) indicated – for both catalysts’ series – that the absorption in the far
 356 UV region (band in the range 200-250 nm) increases when the Cu loading increases, owing to the
 357 charge transfer (CT) transition $O^{2-} (2p) \rightarrow Cu^{2+} (3d)$. Also, an additional broad absorption is detected
 358 at the opposite limit of the visible range, due to the $d \rightarrow d$ transitions in dispersed Cu^{2+} . In the case
 359 of A-CuXSi samples the intensity of both absorptions increases as a function of copper loading, with
 360 a linear trend (see function in Figure S3). In parallel, the maximum of the $d \rightarrow d$ transition tends to
 361 shift to lower wavelengths. This suggests that the nature of copper species is essentially always of the
 362 same nature with increasing loading, the shift of the $d \rightarrow d$ transition being likely due to the decreasing
 363 distance between the dispersed copper centres. In the spectra of I-CuXSi samples, at the highest
 364 loading, the shape of the absorption band due to the $d \rightarrow d$ transitions of dispersed Cu^{2+} ions are
 365 modified, suggesting that an additional absorption grows due to the $d \rightarrow d$ transition in bulk CuO.
 366 This is fully consistent with the appearance of CuO diffraction peaks in the XRD pattern of these
 367 samples. The linear trend between the specific absorption intensities and the copper loading (as
 368 described above for A-catalysts) is verified also for I-Cu2Si and I-Cu5Si but not for I-Cu7.4Si and I-
 369 Cu9.1Si, suggesting a more accentuated heterogeneity of these catalysts in terms of Cu speciation.

370 3.2 Ethanol dehydrogenation

371 Before being tested in the ethanol dehydrogenation reaction, the catalysts were pre-reduced in situ in
372 the reactor. For the sake of completeness, XRD data of pre-reduced catalysts are introduced in Figure
373 S6, showing effective catalyst reduction by production of metallic copper particles for all catalysts
374 (above 2 wt% Cu). Crystal sizes evaluation, carried out on the most intense peak ($2\theta = 35.6^\circ$) by
375 means of the Scherrer equation, indicated the presence of Cu particles around 35 nm for the
376 impregnated catalysts. For aerosol-made catalysts, on the other hand, crystal size evaluation is not
377 possible because the peaks are broad and weak. These pattern features suggest that upon pre-reduction
378 Cu remains highly dispersed throughout the silica spheres at a nanometric scale.

379 In Figure 7, catalytic performances achieved over investigated materials are summarised in terms of
380 ethanol conversion and acetaldehyde selectivity. Over impregnated catalysts, the lowest conversion
381 values are found for I-Cu₂Si, achieving almost 20% at 573 K. The highest conversion performances
382 are reached by I-Cu₅Si, showing already 54% ethanol conversion at 573 K. However, further
383 increasing the Cu loading leads to a marked drop in performance, both in terms of conversion and in
384 terms of selectivity. The drastic drop in selectivity is particularly marked for the catalysts with the
385 highest Cu loading. This appears in line with the characterization survey that indicated a marked
386 aggregation of large CuO (Cu, after reduction) in these high-loading samples prepared by
387 impregnation. Full selectivity is obtained at 473 K and high selectivity (97%) is maintained at higher
388 temperatures for the catalysts with the lowest loading. However, the two catalysts with the highest
389 Cu loading show a marked drop in acetaldehyde selectivity when the temperature was increased to
390 523 K and higher (down to 86% at 673 K). Selectivities to diethyl ether and ethylene are reported in
391 Fig. S7 A.

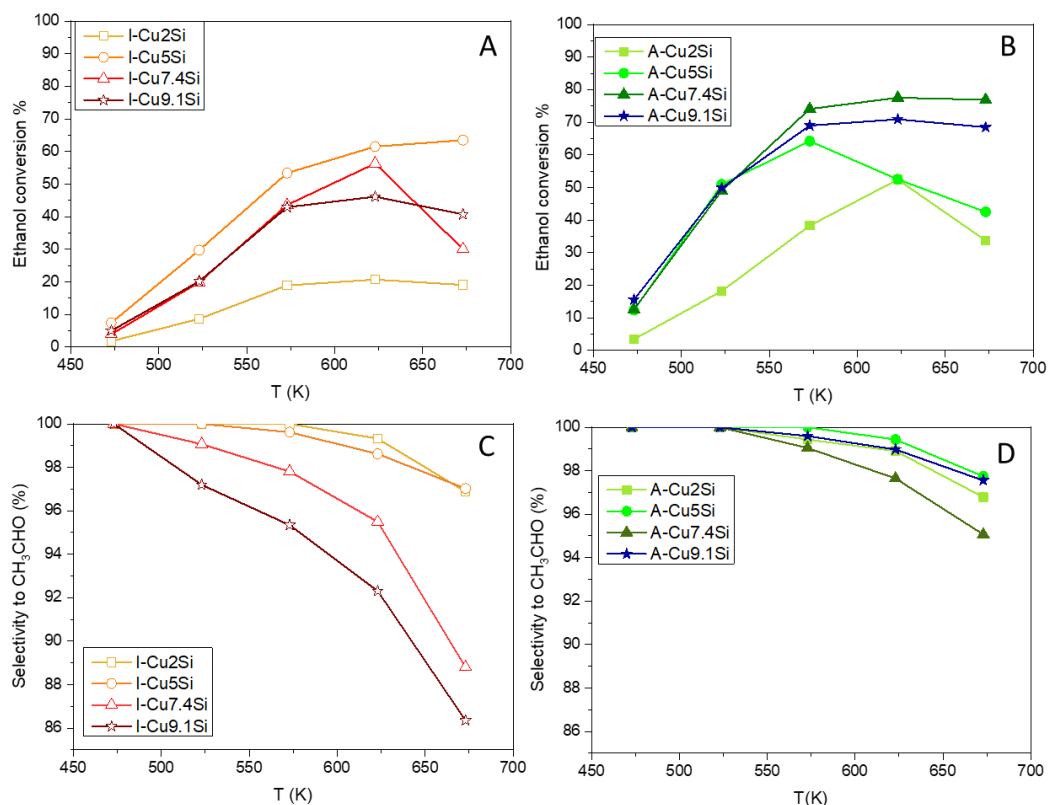


Figure 7: Ethanol conversion (A, C) and acetaldehyde selectivity (B, D) obtained with I-catalysts and A-catalysts, as a function of reaction temperature. Performances are measured by maintaining each reaction temperature constant for 1 hour, from 473 to 673 K.

In comparison with impregnated catalysts, A-CuXSi catalysts reached higher conversion. Performance increases with the Cu loading, up to A-Cu7.4Si, then were slightly lower over A-Cu9.1Si. In particular, at low temperatures aerosol-made catalysts feature almost double catalytic activity when compared to the most performant I-sample, and increased acetaldehyde selectivity with respect to I-catalysts (ESI, Fig. S7, A). These materials reached a plateau in terms of conversion above 573 K (Fig. 7C). Only a slight decrease in the selectivity (from full selectivity to ~97%) to acetaldehyde is observed when increasing the reaction temperature, due to the occurrence of ethanol dehydration to ethylene (ESI, Fig. S7, B).

405 A recent review on the topic allows comparing these catalysts with the state of the art ²⁶, even if it
406 must be stressed that operating conditions often diverge significantly from one study to another. Apart
407 from one Cu/SiO₂ catalyst reported by Zhang et al.⁴⁸ using a relatively high copper amount and a
408 much lower GHSV, the catalysts shown in the present work clearly outcompete all other Cu-silica
409 formulations ²⁶. Beyond silica, looking at Cu-ZnAl₂O₄ as one the most intensively studied catalysts,
410 we can compare with our recent results, obtained in very similar reaction conditions and with similar
411 copper loading ⁵⁰. A-Cu_{7.4}Si shows comparable, or slightly higher, ethanol conversion values in the
412 entire temperature range 573 K-673 K. Furthermore, it seems to be more active at lower temperature
413 (i.e. 49% vs 34% at 523 K).

414 Looking at the catalytic performances shown in Figure 7, for many of the prepared materials i.e. I-
415 Cu_{7.4}Si, I-Cu_{9.1}Si, A-Cu₅Si and A-Cu₂Si a drop in the ethanol conversion values can be seen at the
416 highest temperatures, suggesting that catalyst deactivation is occurring. This is also confirmed
417 looking at conversion values versus time plots (ESI-Fig. S7, C, D) that systematically show a
418 decreasing trend during each temperature dwelling. Previous studies pointed out that both copper
419 sintering and carbon deposition can slow down catalytic ethanol dehydrogenation ⁵¹⁻⁵³. Thus, to shed
420 some light on the possible deactivation causes, spent catalysts were characterized, as discussed in the
421 next paragraph.

422 3.3 Spent catalysts characterization

423 XRD patterns of the catalysts after ethanol conversion experiments are reported in Figure 8. In almost
424 all cases, the peaks of metallic copper (cF4-Cu, PCD ref. 1146504) are detected as the main crystalline
425 phase, together with the appearance of characteristic peaks of Cu₂O (cP6, PCD ref. 1831225) phase.
426 The presence of oxidized Cu species might be due to copper oxidation at the expenses of water arising
427 from ethanol dehydration to ethylene ⁵⁴ or as well to slight oxidation occurring in the presence of air
428 during the handling of the powders. I-catalysts generally reveal a clear bigger crystal size (i.e. 48 and
429 10 nm for I-Cu_{9.1}Si and A-Cu_{9.1}Si) respect to the aerosol made ones.

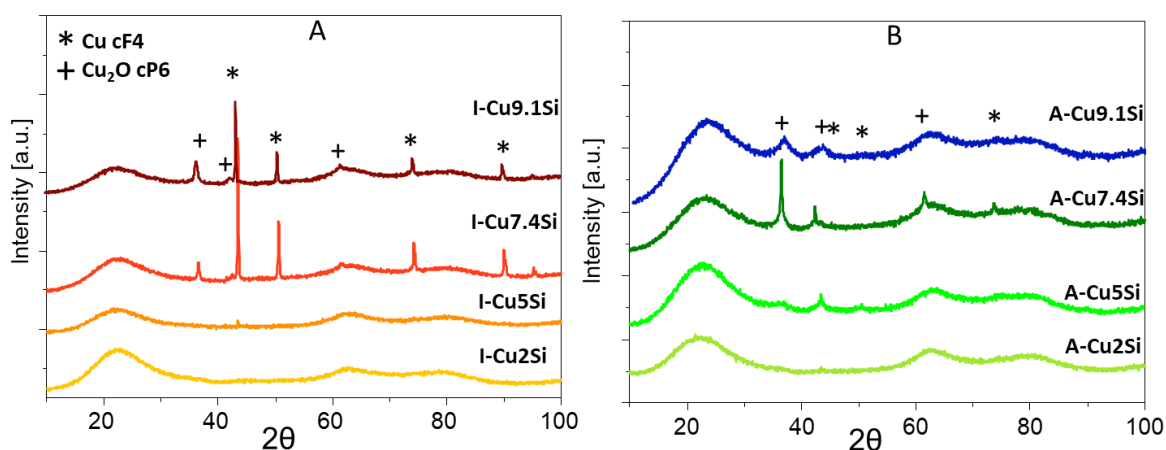


Figure 8: XRD patterns of spent catalysts after catalytic measurements (inset A: I-catalysts; inset B: A-catalysts).

FE-SEM characterization of spent catalysts is reported in Figure 9. In impregnated catalysts, copper-containing particles are visible and appear bigger than in the fresh catalysts. This shows that sintering occurred upon exposure to reaction conditions. Impregnated catalysts show a remarkable high number of copper rich particles that apparently grew in the intraparticle spaces and around small silica particles (Figure 9 A, B). Remarkably, FE-SEM characterization of the A-catalysts (Fig.9 C, D) reveal that upon the same catalytic test conditions, no bright aggregates are detected. The absence of heavy sintering can be interpreted as a sign of a stronger intimacy between the Cu species (nanoparticles) and the silica network, owing to of the aerosol-based method.

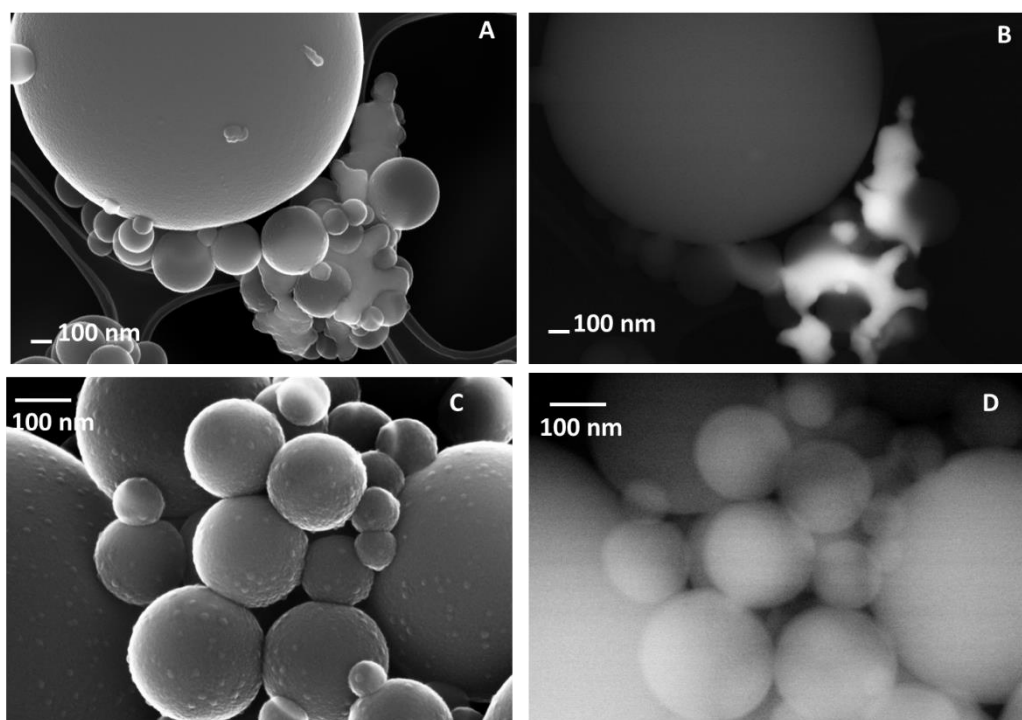


Figure 9: FE-SEM micrographs of spent ICu₅Si (A, B) and Cu₅Si (C, D) catalysts after the catalytic test (473-673 K).

A and C are SE signals. B and D are BSE signals.

TGA analyses for all spent catalysts are reported in Figure S8. TGA of the freshly prepared (not calcined) pristine silica is also provided (Figure S9) as a benchmark to show that the surfactant is fully removed before 700 K, i.e. lower than calcination temperature (823 K). So, the mass losses detected on the spent catalysts account for carbonaceous materials accumulated during reaction. Figure 10, shows the MS signal for $m/z=44$ (CO_2 evolving from carbon oxidation). For all the materials two peaks are found at 530-560 K and 760 K accounting for two different carbon species deposited over catalysts during reaction. The first mass loss can be due to the presence of C-O/C=O containing carbon residue⁵⁵ or amorphous carbon species⁵⁶. It is systematically more important for A-catalysts (that are more active) as compared to I-catalysts (Fig. S8, C). The second CO_2 peak (~760 K) is attributable to the presence of graphitic-like carbon. It is again more important for A-CuXSi, especially at low Cu loading (Fig. S8, C). Fouling by carbonaceous deposit (or coke) is classically linked to the occurrence of acid-catalysed reactions (dehydration, oligomerization, condensation,

etc.). Here, using silica as an “innocent” support, we obtain catalysts that show very low and weak acidity (as shown via NH₃-TPD analyses, see Fig. S10)⁵⁷. In addition, we do not detect the usual dehydration products (e.g. ethylene). Here, coke formation appears to be turnover-dependent: the most active catalysts accumulate more coke. While the usual acid-catalysed path to coke can not be excluded, we surmise that the carbonaceous deposit accumulates through alternative mechanism (e.g. acetaldehyde polymerization).

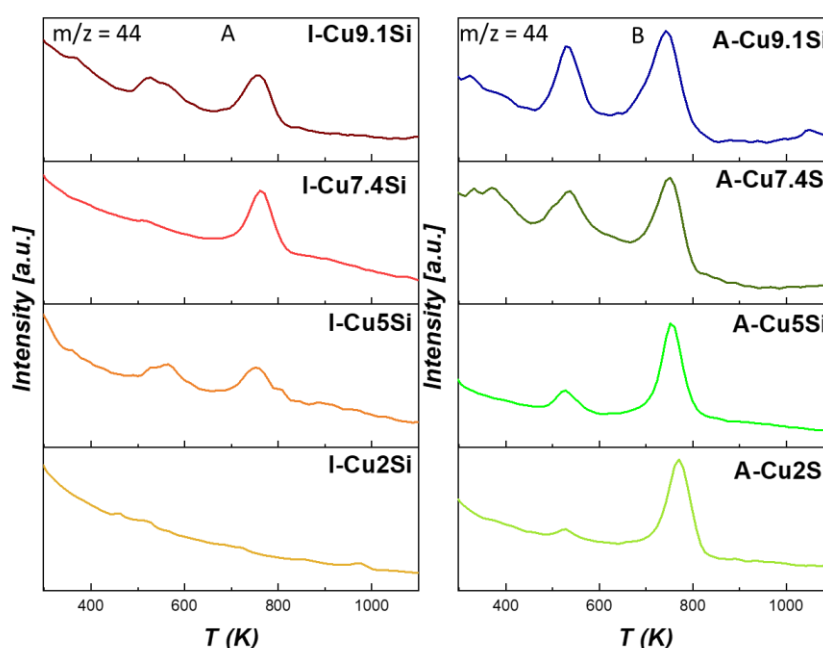


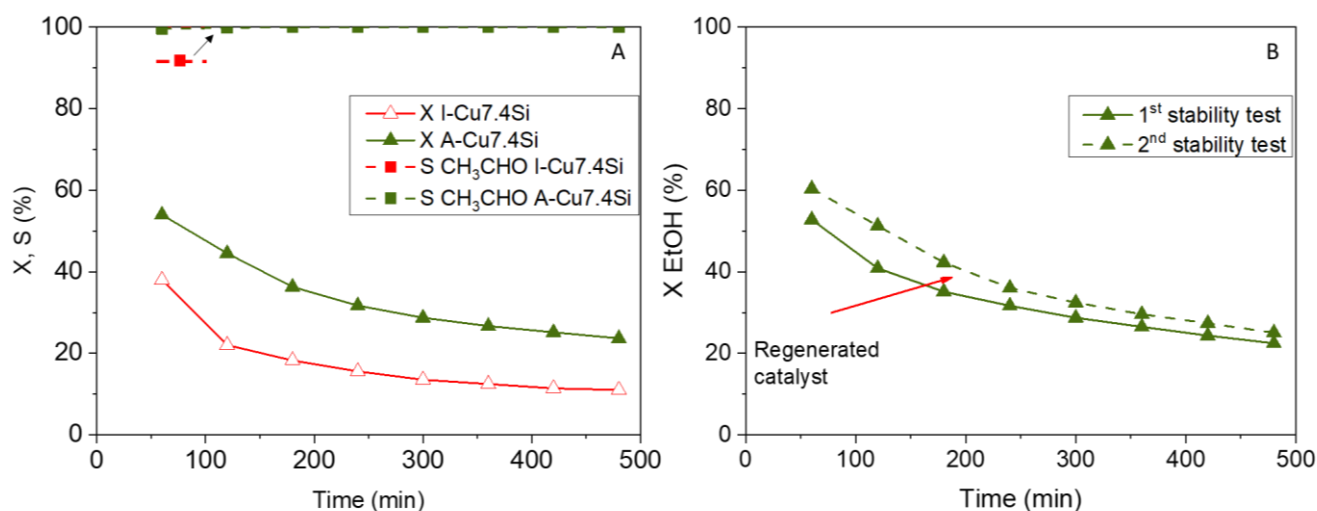
Figure 10: MS monitoring at $m/z = 44$ during the TGA-MS experiment on spent I-catalysts (A) and A-catalysts (B) after catalytic measures (473-673 K).

XPS measurements (Fig. S11, A, B) of the spent catalysts reveal the presence of Cu⁺ and/or Cu⁰ species (consistent with XRD and FE-SEM characterizations) with clear peaks at 932-933 eV (Cu 2p_{3/2} orbitals) and 954 eV (Cu 2p_{1/2} orbitals)⁵⁸. It should be noted that partial re-oxidation can occur during handling which complicates the analysis of the Cu oxidation state in spent catalysts. However, it appears clearly that the Cu/Si ratio tends to decrease after catalytic reaction (Table S3). Causes may be both copper sintering and/or selective coke deposition on copper particles. At this stage of the study is impossible to determine which cause of deactivation is predominant but certainly the

consequence, as suggested in literature^{58,59}, is a reduction of active phase sites at the surface. To further investigate the coke deposition phenomenon (Fig. S11, C, D and Tables S4 and S5), XPS peaks for C1s region was investigated for both fresh and spent catalysts. The surface carbon concentration was found to increase in all spent catalysts, as compared to the fresh counterparts, confirming carbonaceous species deposition. The C1s peak was further decomposed into four contributions: O=C-O (~ 289 eV), C=O/O-C-O (~288-289 eV), C-O (~286 eV), C-(C, H) (~284.8-285 eV)⁶⁰. Comparing fresh and exhausts analyses, the C=O/O-C-O and C-O components remain stable among the different catalysts. Probably underlining the adventitious nature of these species. Components C-(C, H), referred to aliphatic carbon, is more subjected to a variation among the spent catalysts. Considering the components of the C1s peak, it seems that aliphatic carbon species are predominant. This probably confirms the presence of coke with a main C-C-C_n pattern, probably deriving from oligomerized carbonaceous species, in fully agreement with exhaust TGA-MS analyses on spent catalysts. Finally, no satellite peaks due to π - π^* bond (BE > 290 eV) are found, so extended delocalised electrons derived structures (i.e. aromatic species)⁶¹ can be excluded. Thus, we suggest that the carbonaceous deposit that is formed during reaction is mainly composed of amorphous/polymeric coke. Finally, globally looking at the surface carbon concentration over the spent catalysts (ESI, Fig. S12), it seems to be independent of the copper loading for A-catalysts. Instead, over the impregnated ones there is a clear negative trend upon Cu increasing loading. Being that the most active sites are at the border between the support and the nanoparticle, it seems to be present a correlation between Cu crystallites dimensions, catalytic activity and coke deposition. In fact, over impregnated catalysts higher is the loading, and then, proportionally, the crystal size, lower is the coke deposition. Instead, over A-catalysts, where dimensional variation of nanoparticle is less marked, coke deposition seems to be constant.

3.4 Effect of time on stream, deactivation, and rejuvenation

496 The stability of A-Cu_{7.4}Si was tested at 523 K (an industrially relevant operation temperature, where
 497 selectivity for acetaldehyde is high). During 500 min on stream, the selectivity remained complete
 498 (Fig. S13). Conversion declined slowly (from 46% to 36%), indicating progressive deactivation. To
 499 better highlight the difference between I- and A-catalysts, we also applied stability tests at a slightly
 500 higher temperature. A-Cu_{7.4}Si and I-Cu_{7.4}Si catalysts were tested at 573 K for 500 min (Fig. 11 A).
 501 I-Cu_{7.4}Si starts at a conversion of 39%. A-Cu_{7.4}Si reaches a conversion of 55 % (the test was done
 502 with 40 mg of catalyst instead of 50 mg, so that the initial conversion is in the same range for both
 503 catalysts). Both catalysts showed total selectivity to acetaldehyde. Then, conversion slowly declines
 504 with time on stream, tending to a dwelling after 400 minutes. Meanwhile, however, selectivity
 505 remains at 100%. In the course of this stability test, the yield of acetaldehyde is divided by ~4 with I-
 506 Cu_{7.4}Si (down to ~11%), while it is only divided by a factor 2 (down to ~24%) with A-Cu_{7.4}Si.

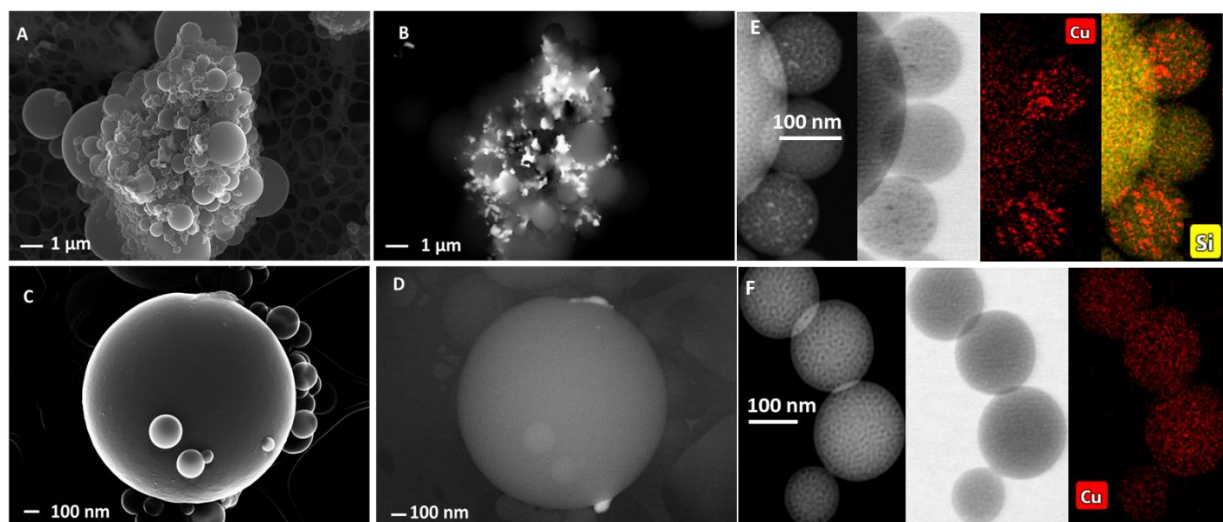


507

508 **Figure 11.** (A) Stability tests over A-Cu_{7.4}Si and I-Cu_{7.4}Si with ethanol conversion and acetaldehyde selectivity
 509 values. (B) stability test and 2nd stability test, on A-Cu_{7.4}Si, after rejuvenation step (mild oxidative treatment) in terms
 510 of ethanol conversion.

511 FE-SEM microphotographs of I-catalyst post stability test (Fig. 12 A, B) reveal a large number of
 512 bright particles (Cu) that aggregate outside of the silica spheres. Thus, it is clear that, although the
 513 stability test has been carried out only at 573 K (i.e. $T < T_{\text{Tamman}}$), segregation of active phase and

514 progressive sintering occurred. As already reported ⁶² atomic migration and crystallite migration
515 (diffusion-controlled processes that occur along the surface, followed by coalescence of
516 atoms/crystallites) are well-known deactivation causes for supported metal nanoparticles. Moreover,
517 pronounced sintering may also occur after prolonged time on stream or/and at high temperatures, as
518 reported by others ¹⁹, generating irreversible deactivation. This effect is very visible on the I-catalyst.
519 On the contrary, after the same test on the A-catalyst (Fig. 12 C, D) only a small number of Cu
520 crystallites is detected. Moreover, looking together at fresh and spent A-catalysts micrographs, no
521 variations in Cu distribution over the support is found. This confirms a slightly reduced mobility of
522 Cu nanoparticles in the catalyst made by aerosol. In fact, it seems reasonable to affirm that A-catalyst
523 shows a higher resistance to coalescence because the Cu nanoparticles are truly embedded in the silica
524 matrix (yet accessible for catalysis, so at least partially pointing towards the external surface of the
525 pores). The fact that Cu species tend to aggregate more on the I-catalyst than on the A-catalyst is
526 unambiguously confirmed by elemental mapping in STEM-EDXS (Fig. 12 E, F) and by XRD (Fig.
527 S14, E, F).



529 **Figure 12:** FE-SEM micrographs obtained on spent catalysts after the stability tests: A (SE signals), B (BSE signals)
530 refer to ICu7.4Si while C (SE signals) and D (BSE) refer to A-Cu7.4Si. STEM images obtained on I-Cu7.4Si (E, dark
531 field, bright field and Cu/Si mapping from left to right) and on A-Cu7.4Si (F, dark field, bright field, and Cu mapping
532 from left to right).

533

534 All in all, both fouling by carbon deposition (ESI Fig. S14, A, B, C, D for TGA-MS), occurring on
535 both catalysts, and more markedly on A-catalysts and Cu sintering, occurring mostly on I-catalysts
536 and only marginally on A-catalysts, can be the main cause of the progressive deactivation. To
537 understand whether the carbon deposition was the major source for the deactivation or not, a
538 rejuvenation was attempted by applying an oxidative treatment (up to 788 K) on the best catalyst (i.e.
539 A-Cu_{7.4}Si) after the stability test. Surprisingly, the 2nd stability test (after the rejuvenation) reached
540 conversion values slightly higher than in the first test (Fig. 11 B) and total selectivity to acetaldehyde
541 was maintained. The origin of this slight increase in activity would require further investigations. Yet,
542 this experiment reveals that the main limiting factor for catalyst stability is the progressive coke
543 deposition ⁶³. However, the carbonaceous deposit can be easily removed using a mild oxidative
544 treatment.

545

546 4. Conclusions

547 Simple and direct synthesis method relying on the aerosol-assisted sol-gel (AASG) process can be
548 utilized to prepare efficient Cu-based catalysts for the non-oxidative dehydrogenation of (bio)ethanol
549 to acetaldehyde. The process leads to spherical and mesostructured silica-based microparticles with
550 highly dispersed Cu species. From a detailed characterization survey, it comes up that preparation
551 methods strongly influence the catalytic properties. In fact, our experimental data suggest that respect
552 to catalysts fabricated by impregnation with one pot made ones is possible to obtain i) smaller
553 nanoparticles, ii) a higher dispersion of Cu at the surface iii) a stronger intimacy between the active
554 phase and the support. A-catalysts present better dehydrogenation performance respect to
555 impregnated ones, in particular at 573 K. Characterization of spent catalysts reveals that deactivation
556 occurs by copper segregation and coke deposition. The main cause of catalyst deactivation seems to
557 be the second one. However, over A-catalysts, where it has been proved that sintering has a limited

558 effect, carbon deposition can be easily reversed with a mild oxidative treatment rejuvenating the
559 catalytic activity.

560

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567

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