

Band Offsets at the Si/SiO₂ Interface from Many-Body Perturbation Theory

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We use many-body perturbation theory, the state-of-the-art method for band gap calculations, to compute the band offsets at the Si/SiO₂ interface. We examine the adequacy of the usual approximations in this context. We show that (i) the separate treatment of band-structure and potential lineup contributions, the latter being evaluated within density-functional theory, is justified, (ii) most plasmon-pole models lead to inaccuracies in the absolute quasiparticle corrections, (iii) vertex corrections can be neglected, (iv) eigenenergy self-consistency is adequate. Our theoretical offsets agree with the experimental ones within 0.3 eV.

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Band offsets (BOs) are among the most important properties of a heterostructure. Their precise knowledge is extremely important to engineer electronic [1] and optoelectronic [2] devices. Various theoretical methods have been used to estimate the BOs [3–5]. Among these, density functional theory (DFT) calculations allow one to study the variation of interface dipoles with the interface structure, without resorting to experimental input [4].

In the DFT approach, the valence band offset (VBO) and the conduction band offset (CBO) are conveniently split into two terms: (V,C)BO = $\Delta E_{v,c}^{DFT} + \Delta V$. The first term $\Delta E_{v,c}^{DFT}$, referred to as the *band-structure contribution*, is defined as the difference between the valence band maxima (VBM) or the conduction band minima (CBM) *relative to the average of the electrostatic potential in each material*. These are obtained from two independent standard bulk calculations on the two materials. The second term ΔV , called the *lineup of the average of the electrostatic potential* across the interface, accounts for all the intrinsic interface effects. It is determined from a supercell calculation with a model interface.

However, the DFT-BOs suffer from two important limitations, namely, the well-known DFT band-gap problem, and the use of an approximate functional to model the exchange-correlation (XC) energy, such as the local density approximation (LDA) or the generalized gradient approximation (GGA). These affect the value of the VBM and the CBM, and consequently the calculated BOs. Orbital-dependent approaches [6] and many-body perturbation-theory (MBPT) within the *GW* approximation [7] have been used to try to overcome these shortcomings [8–10]. While the former may be less computationally demanding, their reliability cannot be assessed a priori [6] in contrast with MBPT. In Refs. [9, 10], the lineup of the potential ΔV was approximated by its DFT value, arguing that it depends primarily on the charge distribution at the interface which is a ground state quan-

tity, hence little affected by many-body effects. Doing so, only the band-structure contribution is modified:

$$\Delta E_{v,c}^{QP} = \Delta E_{v,c}^{DFT} + \Delta(\delta E_{v,c}) \quad (1)$$

where δE_v (δE_c) is the quasiparticle (QP) correction at the VBM (CBM): $\delta E_i = E_i^{QP} - E_i^{DFT}$ for $i = v, c$. It is important to stress that these corrections, which are obtained from bulk calculations, are the only additional ingredients that are required when DFT calculations of the VBO and CBO already exist.

Interestingly, for various semiconductor interfaces, the QP corrections of the band edges are found to be almost the same on both sides [10] leading to $\Delta(\delta E_v) \leq 0.2$ eV in Eq. (1). As a result of this cancellation of errors, DFT is quite successful for the same interfaces [11] with errors ranging from 0.1 to 0.5 eV, despite its limitations mentioned above. This relative success of DFT explains why it has been widely used to predict the VBO for a wide range of interfaces. And, when needed, the CBO was also predicted using a simple scissor operator to correct the band gap to the experimental value. This assumption was further motivated by the fact that MBPT calculations going beyond *GW* by including an approximate vertex correction (*GWT*) showed that the valence-band-edge energy remained at its DFT value for silicon, the whole correction going to the conduction bands [12, 13].

However, when it comes to semiconductor-insulator or insulator-insulator interfaces, it appears that the DFT errors on the VBO can be much more important. For instance, for the Si/SiO₂ interface, the VBO are calculated to be 2.3–3.3 eV [5, 14–16] in noticeable disagreement with the experimental results of 4.3 eV [17]. It seems that, for such systems, the cancellation of errors is not as good, emphasizing the need to go beyond DFT by including QP corrections. The same is probably true for interfaces found in transistors or organic photovoltaics. Before applying such a highly-demanding method in a

predictive way, its quality needs to be assessed for an interface for which the VBO and CBO are well-known experimentally, such as the Si/SiO₂ one.

In this Letter, the effect of QP corrections on the VBO and CBO at the Si/SiO₂ interface is investigated. By performing a quasiparticle self-consistent *GW* (QSGW) calculation on a small interface model, we demonstrate that the line up potential is well described in DFT. For two more realistic models, the BOs calculated within DFT are corrected by computing the QP corrections for bulk Si and SiO₂. The latter are found to be crucial to reach a good agreement with experimental results. Their effect can not be accounted for by a simple scissor operator on top of DFT results. The inclusion of the vertex correction is shown not to affect the *GW* results significantly, whereas QSGW does not improve the results.

Three different interface models are adopted in which ideal β -cristobalite is matched to the Si(100) surface, taking the theoretical value of bulk Si ($a_{Si}=5.48$ Å) as the lattice constant. The models I and II were generated in a previous work by two of us [16] in order to calculate the BOs within DFT. They consist of 11 Si monolayers and up to 10 SiO₂ molecular layers which guarantees well converged BOs. Model III is limited to 8 Si monolayers and 8 SiO₂ molecular layers in order to perform a QSGW calculation which is quite demanding computationally. For bulk Si, the theoretical value a_{Si} is adopted for the lattice constant. For SiO₂, we consider on the one hand the cubic structure (space group $Fd\bar{3}m$) whose relaxed lattice constant is $a_{SiO_2}=7.43$ Å; and, on the other hand, a tetragonal structure (space group $I4_1/amd$) in which two sides are strained to match the Si lattice constant $a_{SiO_2}=\sqrt{2}a_{Si}$ as in the interface models of Ref. [16], while the third is left free to relax reaching a value of $c_{SiO_2}=6.59$ Å. In both structures, the Si-O-Si bond angle is 180°. Hereafter, the cubic and strained structures will be referred to as c-SiO₂ and s-SiO₂, respectively.

All our calculations are performed using the ABINIT package [18]. Only valence electrons are explicitly considered using norm-conserving pseudopotentials [19] to account for core-valence interactions. By coherence with the DFT calculations of the BOs [16], the XC energy is also described within the GGA [20].

At the DFT level, the band gaps E_g^{DFT} for the bulk systems are found to be 0.7 eV for Si, 5.4 eV for c-SiO₂, and 5.1 eV for s-SiO₂ [21]. For the interface models I and II, the VBOs (CBOs) are calculated to be 2.6 and 2.5 eV (1.6 and 1.8 eV), respectively [22].

Three approaches are used to compute the QP corrections. In the first approach (*GW*), the self-energy is calculated self-consistently by updating the eigenenergies in both the dielectric matrices and the Green's functions while keeping the DFT wavefunctions. In the second approach (QSGW), both the eigenenergies and the wavefunctions are updated [24]. In both approaches, the dielectric matrices are computed in the random-phase

approximation (RPA) using the sum over states formulation [25]. In the third approach, a vertex correction Γ is included in both the screening and the self-energy by going beyond the RPA and including XC effects [12], and only the eigenenergies are updated. Details about the DFT-GGA kernel are provided in Ref. [26].

Within *GW*, the frequency dependence of the dynamically screened Coulomb potential W is most often approximated using various plasmon pole models (PPMs) [27–30]. The advantage is not only to reduce the computational load, but also to obtain an analytic expression for the self-energy. In fact, PPMs have proven to be very effective in producing band gaps in good agreement with experiments.

However, while the QP correction to the gap δE_g was found *not* to be very sensitive to the choice of the PPM [31], we observe that δE_v and δE_c may vary from one PPM to another, as reported in Table I for Si and c-SiO₂. In particular, the variation of δE_g with the PPM is more pronounced in c-SiO₂ (up to 0.4 eV difference between the extreme cases) than in Si. Since a precise knowledge of the QP corrections at the band edges is required for the band offsets calculations, *it is necessary to go beyond PPMs* taking explicitly into account the frequency dependence of W .

TABLE I: Quasiparticle corrections (in eV) at the VBM (δE_v), at the CBM (δE_c), and for the band gap (δE_g) for Si and c-SiO₂. The corrections are calculated within *GW* using the PPMs proposed by Hybertsen and Louie (HL) [27], von der Linden and Horsch (vdLH) [28], Godby and Needs (GN) [29], Engel and Farid (EF) [30], and without PPM.

		HL	vdLH	GN	EF	no PPM
Si	δE_v	-0.6	-0.6	-0.4	-0.6	-0.4
	δE_c	+0.1	+0.1	+0.2	+0.1	+0.2
	δE_g	+0.7	+0.7	+0.6	+0.7	+0.6
c-SiO ₂	δE_v	-2.6	-2.5	-2.0	-2.3	-1.9
	δE_c	+1.3	+1.1	+1.5	+1.2	+1.5
	δE_g	+3.9	+3.6	+3.5	+3.5	+3.4

In this work, the explicit frequency dependence is obtained using the deformed contour integration technique [32]. The calculated QP corrections are reported in Table II for Si, c-SiO₂, and s-SiO₂.

The QP gaps can easily be obtained from these results $E_g^{QP} = E_g^{DFT} + \delta E_g$. Within *GW*, $E_g^{QP}=1.3$ eV for Si and 8.8 eV for c-SiO₂. These results are in good agreement with the experimental values of 1.2 and 8.9 eV [33], respectively. While *GW* Γ results (1.3 and 8.5 eV) are very close, the QSGW results (1.5 and 9.5 eV) are overestimated like systematically observed in Ref. [34].

Turning to δE_v and δE_c , the comparison of our results with previous calculations is not always straightforward. In most of the cases, the focus is on the band gap and

TABLE II: Quasiparticle corrections (in eV) at the VBM (δE_v), at the CBM (δE_c), and for the band gap (δE_g) for Si, c-SiO₂, and s-SiO₂. The corrections are calculated within *GW*, *GWT*, and *QSGW*.

	Si			c-SiO ₂			s-SiO ₂		
	<i>GW</i>	<i>GWT</i>	<i>QSGW</i>	<i>GW</i>	<i>GWT</i>	<i>QSGW</i>	<i>GW</i>	<i>GWT</i>	<i>QSGW</i>
δE_v	-0.4	+0.1	-0.6	-1.9	-1.3	-2.8	-1.9	-1.3	-2.8
δE_c	+0.2	+0.7	+0.2	+1.5	+1.8	+1.3	+1.4	+1.8	+1.1
δE_g	+0.6	+0.6	+0.8	+3.4	+3.1	+4.1	+3.3	+3.1	+3.9

the QP corrections are not given explicitly (the VBM is set to zero after correction). Besides, δE_v and δE_c are very sensitive to the degree of convergence reached (the QP gap converges much faster than the QP corrections), not to mention the effect of the PPM (see Table I). For Si, our *GW* and *GWT* results compare quite well with those of Fleszar and Hanke [13] which also do not rely on any PPM [35]. No such comparison can be made for c-SiO₂. Note however that the QP corrections for c-SiO₂ and s-SiO₂ differ by up to 0.2 eV, indicating a dependence of δE_v and δE_c on the strain. This is at variance with the findings of Ref. [36] in the case of Si under isotropic strain. A possible explanation for this difference may be that the strain is not isotropic in s-SiO₂ [21]. Finally, note that the PPM proposed by Godby and Needs [29] leads to QP corrections (see Table I) in excellent agreement with those of the contour deformation method, at variance with the other PPMs. This finding might be generalized after proper investigation.

While *GW* and *QSGW* lead to a lowering of the VBM of Si (slightly larger for *QSGW*) compared to the DFT result, the inclusion of the vertex correction brings it back to roughly its original value with a small shift of 0.1 eV, all the QP correction being on the conduction band. A similar result was also found previously [12, 13] giving some motivation to the use of a scissor operator to compute the CBO within DFT. For SiO₂, our results are very different. First, the VBM is also raised when going from *GW* to *GWT*, but it definitely does not reach the DFT level back. This indicates the recovery of the DFT VBM with *GWT* is a coincidence in Si. And, it definitely rules out the use of a simple scissor operator for the computation of the BOs. Second, the lowering of the VBM in *QSGW* is much larger than in *GW* (by about 0.9 eV). In fact, it seems that the systematical overestimation of the band gap in *QSGW* originates essentially from a too strong lowering of the VBM. Note that the increase of the CBM is slightly lower in *QSGW* than in *GW*.

In order to compute the QP corrections to the BOs, the many-body effects on ΔV also need to be investigated. This is done by comparing the electronic density and the resulting ΔV calculated within DFT and *QSGW* for model III. In Fig. 1(a), the difference between the planar

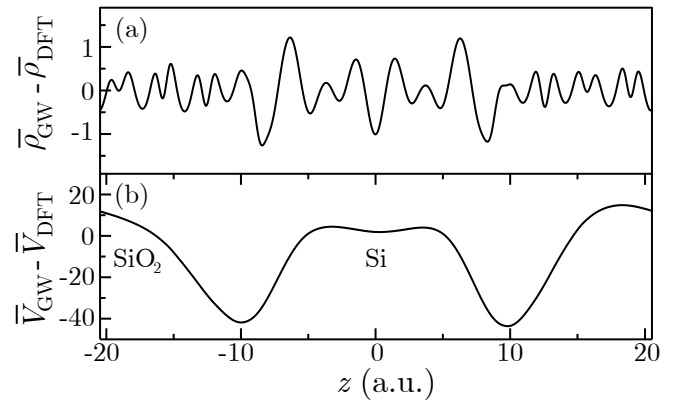


FIG. 1: Difference between DFT and *QSGW* calculations for model III of (a) the planar average of the electronic density and (b) the macroscopic average of the local potential. The density is expressed in $me/a.u.$, and the potential in meV .

nar average of the DFT electronic density $\bar{\rho}_{DFT}(z)$ and the *QSGW* one $\bar{\rho}_{GW}(z)$ is reported. The *QSGW* result differs only slightly from DFT, with a maximum change of about 1 $me/a.u.$ in the interface region. The difference between the *QSGW* and DFT macroscopic average of the local potential $\bar{V}_{GW} - \bar{V}_{DFT}$, which is reported in Fig. 1(b), is about ~ 45 meV in the interface region, and less than 12 meV in the bulk regions. This gives rise to a net difference of ~ 20 meV in the lineup of the potential ΔV . Such a small difference in ΔV suggests that the interfacial charge density and, consequently, the associated dipole moments are well described within DFT. This justifies the assumption that the line up potential can be taken to be the same as in DFT.

Finally, using Eq. (1), the band offsets can be computed within MBPT at the *GW*, *GWT*, and *QSGW* levels. Our results are reported in Table III and compared with the experimental ones. Within *GW* the agreement is excellent for both the VBO and CBO (less than 0.3 eV difference). The effect of the vertex correction is less than 0.1 eV on the BOs. This results from a cancellation of the effects on each side of the interface. At this stage, we cannot say whether this result can be generalized to any interface. In contrast, the effect of quasiparticle self-consistency is more pronounced. Due to the large lowering of the VBM of SiO₂, the VBO is increased by 0.7 eV compared to *GW*, leading to an overestimation by up to 0.5 eV. At the same time, the CBO is slightly smaller than in *GW* leading to an underestimation by up to 0.6 eV. So that *QSGW* is the worst of the three approaches.

In summary, we have investigated the band offsets at the Si/SiO₂ interface using many-body perturbation theory. Starting from the BOs obtained within DFT for two model interfaces, the QP corrections have been computed within eigenenergy self-consistent *GW* and *GWT*, and quasiparticle self-consistent *GW* taking the frequency dependence of the screened potential explicitly into account

TABLE III: Quasiparticle band offsets (eV) for cubic and strained SiO₂ using *GW*, *GWT*, and *QSGW*.

	Model	DFT	cubic			strained			Expt.
			<i>GW</i>	<i>GWT</i>	<i>QSGW</i>	<i>GW</i>	<i>GWT</i>	<i>QSGW</i>	
VBO	I	2.6	4.1	4.0	4.8	4.1	4.0	4.8	4.3
	II	2.5	4.0	3.9	4.7	4.0	3.9	4.7	
CBO	I	1.6	2.9	2.7	2.7	2.8	2.7	2.5	3.1
	II	1.8	3.1	2.9	2.9	3.0	2.9	2.7	

(which is more reliable than plasmon pole models). The *GW* corrections, which differ significantly from what can be obtained using a scissor operator, produce BOs in excellent agreement with experiment. While the BOs obtained with and without vertex correction do not differ significantly, those resulting from *QSGW* present a larger deviation from experiments. These findings allow us to recommend the use of eigenenergy self-consistent *GW*, which is less demanding. Now that the quality of the procedure has been assessed, it can be used in a predictive way for heterojunctions of high technological interest such as those in transistors or organic photovoltaics.

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- [1] G.D. Wilk, R.M. Wallace, and J.M. Anthony, J. Appl. Phys. **89**, 5243 (2001).
- [2] A.C. Morteani, *et al.*, Phys. Rev. Lett. **92**, 247402 (2004).
- [3] C. Tejedor and F. Flores, J. Phys. C **11**, L19 (1977); W.A. Harrison, Electronic Structure and the Properties of Solids (Freeman, San Francisco, 1980), p. 253; J. Tersoff, Phys. Rev. B **30**, 4874 (1984); J. Robertson, J. Vac. Sci. Technol. B **18**, 1785 (2000); P.W. Peacock and J. Robertson, Appl. Phys. **92**, 4712 (2002).
- [4] C.G. Van de Walle and R.M. Martin, Phys. Rev. B **34**, 5621 (1986).
- [5] T. Yamasaki *et al.*, Phys. Rev. B **63**, 115314 (2001).
- [6] S. Kümmel and L. Kronik, Rev. Mod. Phys. **80**, 3 (2008).
- [7] L. Hedin, Phys. Rev. **139**, A796 (1965).
- [8] F. Devynck, F. Giustino, P. Broqvist, and A. Pasquarello, Phys. Rev. B **76**, 075351 (2007).
- [9] S. B. Zhang *et al.*, Solid State Commun. **66**, 585 (1988).
- [10] X. Zhu and S. G. Louie, Phys. Rev. B **43**, 14142 (1991).
- [11] C.G. Van de Walle and R.M. Martin, Phys. Rev. B **35**, 8154 (1987).
- [12] R. Del Sole, L. Reining, and R.W. Godby, Phys. Rev. B **49**, 8024 (1994).
- [13] A. Fleszar and W. Hanke, Phys. Rev. B **56**, 10228 (1997).
- [14] M. Watarai, J. Nakamura, and A. Natori, Phys. Rev. B **69**, 035312 (2004).
- [15] B.R. Tuttle, Phys. Rev. B **70**, 125322 (2004).
- [16] F. Giustino and A. Pasquarello, Surf. Sci. **586** 183 (2005).
- [17] J.W. Keister *et al.*, J. Vac. Sci. Technol. B **17**, 1831 (1999); V.V. Afanas'ev, M. Houssa, and A. Stesman, Appl. Phys. Lett. **78**, 3073 (2001).
- [18] X. Gonze *et al.*, Zeit. Kristall. **220**, 558 (2005).
- [19] N. Troullier and J.L. Martins, Phys. Rev. B **43**, 1993 (1991).
- [20] J.P. Perdew, *et al.*, Phys. Rev. B **46**, 6671 (1992).
- [21] For Si, the gap is indirect with the VBM located at Γ and the CBM approximated by the X point; while c-SiO₂ has a direct gap at Γ . The lowering of the symmetry in s-SiO₂ results in a splitting of the threefold-degenerate state at Γ into two top valence states, the highest corresponding to the p_z orbital, and the other (twofold-degenerate) to the p_x and p_y orbitals. These two states are separated by 0.27 eV, accounting basically for the difference in the band gap between the two SiO₂ structures.
- [22] In Ref. [16], the BOs were obtained directly from the local density of states in the supercell. The difference between the two models was attributed to different interface dipoles. Interestingly, the calculated BOs are almost identical to those of more sophisticated models in which the SiO₂ side is taken to be disordered [23], justifying further comparison with experiments in which the silica is amorphous. Note that the band gaps calculated directly in the interface models are slightly smaller than those of the bulk calculations probably due to the lowering of the symmetry resulting from the atomic relaxation.
- [23] F. Giustino, A. Bongiorno, and A. Pasquarello Appl. Phys. Lett. **86**, 192901 (2005).
- [24] S.V. Faleev, M. van Schilfgaarde, T. Kotani, Phys. Rev. Lett. **93**, 126406 (2004).
- [25] L. Adler, Phys. Rev. **126**, 413 (1962); N. Wiser, *ibid.* **129**, 62 (1963).
- [26] A. Dal Corso, S. Baroni, and R. Resta Phys. Rev. B **49**, 5323 (1994).
- [27] M.S. Hybertsen and S.G. Louie, Phys. Rev. B **34**, 5390 (1986).
- [28] W. von der Linden and P. Horsch, Phys. Rev. B **37**, 8351 (1988).
- [29] R.W. Godby and R.J. Needs, Phys. Rev. Lett. **62**, 1169 (1989).
- [30] G.E. Engel and B. Farid, Phys. Rev. B **47**, 15931 (1993).
- [31] B. Arnaud and M. Alouani, Phys. Rev. B **62**, 4464 (2000).
- [32] S. Lebègue, B. Arnaud, M. Alouani and P. E. Bloechl, Phys. Rev. B **67**, 155208 (2003).
- [33] T.H. Di Stefano and D.E. Eastman, Solid Stat. Comm. **9**, 2259 (1971).
- [34] M. Shishkin, M. Marsman, and G. Kresse, Phys. Rev. Lett. **99**, 246403 (2007).
- [35] The slight differences are probably due to the use of a different XC energy functional at the DFT level (GGA in our case and LDA in the work of Fleszar and Hanke).
- [36] X. Zhu, S. Fahy, and S.G. Louie, Phys. Rev. B **39**, 7840 (1989).