"Band Offsets at the Si/SiO2 Interface from Many-Body Perturbation Theory."

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ABSTRACT

We use many-body perturbation theory, the state-of-the-art method for band-gap calculations, to compute the band offsets at the Si/SiO₂ interface. We examine the adequacy of the usual approximations in this context. We show that (i) the separate treatment of band structure and potential lineup contributions, the latter being evaluated within density-functional theory, is justified, (ii) most plasmon-pole models lead to inaccuracies in the absolute quasiparticle corrections, (iii) vertex corrections can be neglected, and (iv) eigenenergy self-consistency is adequate. Our theoretical offsets agree with the experimental ones within 0.3 eV.

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Band Offsets at the Si/SiO₂ Interface from Many-Body Perturbation Theory

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We use many-body perturbation theory, the state-of-the-art method for band-gap calculations, to compute the band offsets at the Si/SiO₂ interface. We examine the adequacy of the usual approximations in this context. We show that (i) the separate treatment of band structure and potential lineup contributions, the latter being evaluated within density-functional theory, is justified, (ii) most plasmon-pole models lead to inaccuracies in the absolute quasiparticle corrections, (iii) vertex corrections can be neglected, and (iv) eigenenergy self-consistency is adequate. Our theoretical offsets agree with the experimental ones within 0.3 eV.

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Band offsets (BOs) are among the most important properties of a heterostructure. Their precise knowledge is extremely important to engineer electronic [1] and optoelectronic [2] devices. Various theoretical methods have been used to estimate the BOs [3–5]. Among these, density-functional theory (DFT) calculations allow one to study the variation of interface dipoles with the interface structure, without resorting to experimental input [4].

In the DFT approach, the valence band offset (VBO) and the conduction band offset (CBO) are conveniently split into two terms: \( V(C)\text{BO} = \Delta E_{v,c}^{\text{DFT}} + \Delta V \). The first term \( \Delta E_{v,c}^{\text{DFT}} \), referred to as the band structure contribution, is defined as the difference between the valence band maxima (VBM) or the conduction band minima (CBM) relative to the average of the electrostatic potential in each material. These are obtained from two independent standard bulk calculations on the two materials. The second term \( \Delta V \), called the lineup of the average of the electrostatic potential across the interface, accounts for all the intrinsic interface effects. It is determined from a supercell calculation with a model interface.

However, the DFT-BOs suffer from two important limitations, namely, the well-known DFT band-gap problem, and the use of an approximate functional to model the exchange-correlation (XC) energy, such as the local density approximation (LDA) or the generalized gradient approximation (GGA). These affect the value of the VBM and the CBM, and consequently the calculated BOs. Orbital-dependent approaches [6] and many-body perturbation-theory (MBPT) within the GW approximation [7] have been used to try to overcome these shortcomings [8–10]. While the former may be less computationally demanding, their reliability cannot be assessed a priori [6] in contrast with MBPT. In Refs. [9,10], the lineup of the potential \( \Delta V \) was approximated by its DFT value, arguing that it depends primarily on the charge distribution at the interface which is a ground state quantity, hence little affected by many-body effects. Doing so, only the band structure contribution is modified:

\[
\Delta E_{i}^{\text{QP}} = \Delta E_{i}^{\text{DFT}} + \delta E_{i},
\]

where \( \delta E_{i} \) (\( \delta E_{c} \)) is the quasiparticle (QP) correction at the VBM (CBM): \( \delta E_{i} = E_{i}^{\text{QP}} - E_{i}^{\text{DFT}} \) for \( i = v, c \). It is important to stress that these corrections, which are obtained from bulk calculations, are the only additional ingredients that are required when DFT calculations of the VBO and CBO already exist.

Interestingly, for various semiconductor interfaces, the QP corrections of the band edges are found to be almost the same on both sides [10] leading to \( \Delta(\delta E_{i}) \leq 0.2 \text{ eV} \) in Eq. (1). As a result of this cancellation of errors, DFT is quite successful for the same interfaces [11] with errors ranging from 0.1 to 0.5 eV, despite its limitations mentioned above. This relative success of DFT explains why it has been widely used to predict the VBO for a wide range of interfaces. And, when needed, the CBO was also predicted using a simple scissor operator to correct the band gap to the experimental value. This assumption was further motivated by the fact that MBPT calculations going beyond GW by including an approximate vertex correction (GWf) showed that the valence-band-edge energy remained at its DFT value for silicon, the whole correction going to the conduction bands [12,13].

However, when it comes to semiconductor-insulator or insulator-insulator interfaces, it appears that the DFT errors on the VBO can be much more important. For instance, for the Si/SiO₂ interface, the VBO are calculated to be 2.3–3.3 eV [5,14–16] in noticeable disagreement with the experimental results of 4.3 eV [17]. It seems that, for such systems, the cancellation of errors is not as good, emphasizing the need to go beyond DFT by including QP corrections. The same is probably true for interfaces found in transistors or organic photovoltaics. Before applying such a highly demanding method in a predictive way, its
quality needs to be assessed for an interface for which the VBO and CBO are well known experimentally, such as the Si/SiO2 one.

In this Letter, the effect of QP corrections on the VBO and CBO at the Si/SiO2 interface is investigated. By performing a quasiparticle self-consistent GW (QSGW) calculation on a small interface model, we demonstrate that the lineup potential is well described in DFT. For two more realistic models, the BOs calculated within DFT are corrected by computing the QP corrections for bulk Si and SiO2. The latter are found to be crucial to reach a good agreement with experimental results. Their effect cannot be accounted for by a simple scissor operator on top of DFT results. The inclusion of the vertex correction is shown not to affect the GW results significantly, whereas QSGW does not improve the results.

Three different interface models are adopted in which ideal β cristobalite is matched to the Si(100) surface, taking the theoretical value of bulk Si ($a_{Si} = 5.48$ Å) as the lattice constant. Models I and II were generated in a previous work by two of us [16] in order to calculate the BOs within DFT. They consist of 11 Si monolayers and up to 10 SiO2 molecular layers which guarantees well converged BOs. Model III is limited to 8 Si monolayers and 8 SiO2 molecular layers in order to perform a QSGW calculation which is quite demanding computationally. For bulk Si, the theoretical value $a_{Si}$ is adopted for the lattice constant. For SiO2, we consider on the one hand the cubic structure (space group Fd3m) whose relaxed lattice constant is $a_{SiO2} = 7.43$ Å, and, on the other hand, a tetragonal structure (space group I41/amd) in which two sides are strained to match the Si lattice constant $a_{SiO2} = \sqrt{2}a_{Si}$ as in the interface models of Ref. [16], while the third is left free to relax reaching a value of $c_{SiO2} = 6.59$ Å. In both structures, the Si-O-Si bond angle is 180°. Hereafter, the cubic and strained structures will be referred to as c-SiO2 and s-SiO2, respectively.

All our calculations are performed using the ABINIT package [18]. Only valence electrons are explicitly considered using norm-conserving pseudopotentials [19] to account for core-valence interactions. By coherence with the DFT calculations of the BOs [16], the XC energy is also described within the GGA [20].

At the DFT level, the band gaps $E_{g}^{DFT}$ for the bulk systems are found to be 0.7 eV for Si, 5.4 eV for c-SiO2, and 5.1 eV for s-SiO2 [21]. For the interface models I and II, the VBOs (CBOs) are calculated to be 2.6 and 2.5 eV (1.6 and 1.8 eV), respectively [22].

Three approaches are used to compute the QP corrections. In the first approach (GW), the self-energy is calculated self-consistently by updating the eigenenergies in both the dielectric matrices and the Green’s functions while keeping the DFT wave functions. In the second approach (QSGW), both the eigenenergies and the wave functions are updated [24]. In both approaches, the dielectric matrices are computed in the random-phase approximation (RPA) using the sum over states formulation [25]. In the third approach, a vertex correction $\Gamma$ is included in both the screening and the self-energy by going beyond the RPA and including XC effects [12], and only the eigenenergies are updated. Details about the DFT-GGA kernel are provided in Ref. [26].

Within GW, the frequency dependence of the dynamically screened Coulomb potential $W$ is most often approximated using various plasmon-pole models (PPMs) [27–30]. The advantage is not only to reduce the computational load, but also to obtain an analytic expression for the self-energy. In fact, PPMs have proven to be very effective in producing band gaps in good agreement with experiments.

However, while the QP correction to the gap $\delta E_g$ was found not to be very sensitive to the choice of the PPM [31], we observe that $\delta E_v$ and $\delta E_e$ may vary from one PPM to another, as reported in Table I for Si and c-SiO2. In particular, the variation of $\delta E_g$ with the PPM is more pronounced in c-SiO2 (up to 0.4 eV difference between the extreme cases) than in Si. Since a precise knowledge of the QP corrections at the band edges is required for the band offsets calculations, it is necessary to go beyond PPMs taking explicitly into account the frequency dependence of $W$.

In this Letter, the explicit frequency dependence is obtained using the deformed contour integration technique [32]. The calculated QP corrections are reported in Table II for Si, c-SiO2, and s-SiO2.

#### Table I. Quasiparticle corrections (in eV) at the VBM ($\delta E_v$), at the CBM ($\delta E_e$), and for the band gap ($\delta E_g$) for Si and c-SiO2. The corrections are calculated within GW using the PPMs proposed by Hybertsen and Louie (HL) [27], von der Linden and Horsch (vdLH) [28], Godby and Needs (GN) [29], Engel and Fardil (EF) [30], and without PPM.

<table>
<thead>
<tr>
<th></th>
<th>HL</th>
<th>vdLH</th>
<th>GN</th>
<th>EF</th>
<th>no PPM</th>
</tr>
</thead>
<tbody>
<tr>
<td>Si</td>
<td>$\delta E_v$</td>
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<td>-0.6</td>
<td>-0.4</td>
<td>-0.6</td>
</tr>
<tr>
<td></td>
<td>$\delta E_e$</td>
<td>+0.1</td>
<td>+0.1</td>
<td>+0.2</td>
<td>+0.1</td>
</tr>
<tr>
<td></td>
<td>$\delta E_g$</td>
<td>+0.7</td>
<td>+0.7</td>
<td>+0.6</td>
<td>+0.7</td>
</tr>
<tr>
<td>c-SiO2</td>
<td>$\delta E_v$</td>
<td>-2.6</td>
<td>-2.5</td>
<td>-2.0</td>
<td>-2.3</td>
</tr>
<tr>
<td></td>
<td>$\delta E_e$</td>
<td>+1.3</td>
<td>+1.1</td>
<td>+1.5</td>
<td>+1.2</td>
</tr>
<tr>
<td></td>
<td>$\delta E_g$</td>
<td>+3.9</td>
<td>+3.6</td>
<td>+3.5</td>
<td>+3.5</td>
</tr>
</tbody>
</table>

#### Table II. Quasiparticle corrections (in eV) at the VBM ($\delta E_v$), at the CBM ($\delta E_e$), and for the band gap ($\delta E_g$) for Si, c-SiO2, and s-SiO2. The corrections are calculated within GW, GWT, and QSGW.

<table>
<thead>
<tr>
<th></th>
<th>Si</th>
<th>c-SiO2</th>
<th>s-SiO2</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>$\delta E_v$</td>
<td>$\delta E_e$</td>
<td>$\delta E_g$</td>
</tr>
<tr>
<td>GW</td>
<td>-0.4</td>
<td>-0.6</td>
<td>-1.3</td>
</tr>
<tr>
<td>GWT</td>
<td>+0.1</td>
<td>+0.2</td>
<td>+0.7</td>
</tr>
<tr>
<td>QSGW</td>
<td>-0.6</td>
<td>+0.8</td>
<td>+3.4</td>
</tr>
<tr>
<td>GW</td>
<td>-0.6</td>
<td>+0.8</td>
<td>+3.4</td>
</tr>
<tr>
<td>GWT</td>
<td>+0.6</td>
<td>+0.8</td>
<td>+3.4</td>
</tr>
<tr>
<td>QSGW</td>
<td>-1.3</td>
<td>+1.8</td>
<td>+3.1</td>
</tr>
<tr>
<td>2008</td>
<td>-1.9</td>
<td>+3.3</td>
<td>+3.1</td>
</tr>
</tbody>
</table>
The QP gaps can easily be obtained from these results $E^\text{QP}_g = E^\text{DFT}_g + \delta E_g$. Within GW, $E^\text{QP}_g = 1.3$ eV for Si and 8.8 eV for c-SiO$_2$. These results are in good agreement with the experimental values of 1.2 and 8.9 eV [33], respectively. While GW$^\text{L}$ results (1.3 and 8.5 eV) are very close, the QSGW results (1.5 and 9.5 eV) are overestimated like systematically observed in Ref. [34].

Turning to $\delta E_v$ and $\delta E_c$, the comparison of our results with previous calculations is not always straightforward. In most of the cases, the focus is on the band gap and the QP corrections are not given explicitly (the VBM is set to zero after correction). Besides, $\delta E_v$ and $\delta E_c$ are very sensitive to the degree of convergence reached (the QP gap converges much faster than the QP corrections), not to mention the effect of the PPM (see Table I). For Si, our GW and GW$^\text{L}$ results compare quite well with those of Fleszar and Hanke [13] which also do not rely on any PPM [35]. No such comparison can be made for c-SiO$_2$. Note, however, that the QP corrections for c-SiO$_2$ and s-SiO$_2$ differ by up to 0.2 eV, indicating a dependence of $\delta E_v$ and $\delta E_c$ on the strain. This is at variance with the findings of Ref. [36]. In the case of Si under isotropic strain. A possible explanation for this difference may be that the strain is not isotropic in s-SiO$_2$. Finally, note that the PPM proposed by Godby and Needs [29] leads to QP corrections (see Table I) in excellent agreement with those of the contour deformation method, at variance with the other PPMs. This finding might be generalized after proper investigation.

While GW and QSGW lead to a lowering of the VBM of Si (slightly larger for QSGW) compared to the DFT result, the inclusion of the vertex correction brings it back to roughly its original value with a small shift of 0.1 eV, all the QP correction being on the conduction band. A similar result was also found previously [12,13] giving some motivation to the use of a scissor operator to compute the CBO within DFT. For SiO$_2$, our results are very different. First, the VBM is also raised when going from GW to GWT, but it definitely does not reach the DFT level back. This indicates the recovery of the DFT VBM with GWT is a coincidence in Si. And, it definitely rules out the use of a simple scissor operator for the computation of the QPs. Second, the lowering of the VBM in QSGW in much larger than in GW (by about 0.9 eV). In fact, it seems that the systematical overestimation of the band gap in QSGW originates essentially from a too strong lowering of the VBM. Note that the increase of the CBM is slightly lower in QSGW than in GW.

In order to compute the QP corrections to the QVs, the many-body effects on $\Delta V$ also need to be investigated. This is done by comparing the electronic density and the resulting $\Delta V$ calculated within DFT and QSGW for model III. In Fig. 1(a), the difference between the planar average of the DFT electronic density $\tilde{n}_\text{DFT}(z)$ and the QSGW one $\tilde{n}_\text{GW}(z)$ is reported. The QSGW result differs only slightly from DFT, with a maximum change of about 1 me/\text{a.u.} in the interface region. The difference between the QSGW and DFT macroscopic average of the local potential $\tilde{V}_\text{GW} - \tilde{V}_\text{DFT}$, which is reported in Fig. 1(b), is about $\sim 45$ meV in the interface region, and less than 12 meV in the bulk regions. This gives rise to a net difference of $\sim 20$ meV in the lineup of the potential $\Delta V$. Such a small difference in $\Delta V$ suggests that the interfacial charge density and, consequently, the associated dipole moments are well described within DFT. This justifies the assumption that the lineup potential can be taken to be the same as in DFT.

Finally, using Eq. (1), the band offsets can be computed within MBPT at the GW, GWT, and QSGW levels. Our results are reported in Table III and compared with the experimental ones. Within GW the agreement is excellent for both the VBO and CBO (less than 0.3 eV difference). The effect of the vertex correction is less than 0.1 eV on the BOs. This results from a cancellation of the effects on each side of the interface. At this stage, we cannot say whether this result can be generalized to any interface. In contrast, the effect of quasiparticle self-consistency is more pronounced. Because of the large lowering of the VBM of SiO$_2$, the VBO is increased by 0.7 eV compared with GW, leading to an overestimation by up to 0.5 eV. At the same time, the CBO is slightly smaller than in GW leading to an underestimation by up to 0.6 eV. So that QSGW is the worst of the three approaches.

### TABLE III. Quasiparticle band offsets (eV) for cubic and strained SiO$_2$ using GW, GWT, and QSGW.

<table>
<thead>
<tr>
<th></th>
<th>Cubic</th>
<th>Strained</th>
</tr>
</thead>
<tbody>
<tr>
<td>Model</td>
<td>DFT</td>
<td>$GW^\text{L}$</td>
</tr>
<tr>
<td>VBO</td>
<td>1.6</td>
<td>2.4</td>
</tr>
<tr>
<td>CBO</td>
<td>1.8</td>
<td>2.9</td>
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<td></td>
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</tbody>
</table>
In summary, we have investigated the band offsets at the Si/SiO₂ interface using many-body perturbation theory. Starting from the BOs obtained within DFT for two model interfaces, the QP corrections have been computed within eigenenergy self-consistent GW and GWT, and quasiparticle self-consistent GW taking the frequency dependence of the screened potential explicitly into account (which is more reliable than plasmon-pole models). The GW corrections, which differ significantly from what can be obtained using a scissor operator, produce BOs in excellent agreement with experiment. While the BOs obtained with and without vertex correction do not differ significantly, those resulting from QSGW present a larger deviation from experiments. These findings allow us to recommend the use of eigenenergy self-consistent GW, which is less demanding. Now that the quality of the procedure has been assessed, it can be used in a predictive way for heterojunctions.

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[21] For Si, the gap is indirect with the VBM located at Γ and the CBM approximated by the X point, while c-SiO₂ has a direct gap at Γ. The lowering of the symmetry in c-SiO₂ results in a splitting of the threefold-degenerate state at Γ into two top valence states, the highest corresponding to the pₓ and orbital, and the other (twofold-degenerate) to the pₓ and pᵧ orbitals. These two states are separated by 0.27 eV, accounting basically for the difference in the band gap between the two SiO₂ structures.
[22] In Ref. [16], the BOs were obtained directly from the local density of states in the supercell. The difference between the two models was attributed to different interface dipoles. Interestingly, the calculated BOs are almost identical to those of more sophisticated models in which the SiO₂ side is taken to be disordered [23], justified by further comparison with experiments in which the silica is amorphous. Note that the band gaps calculated directly in the interface models are slightly smaller than those of the bulk calculations probably due to the lowering of the symmetry resulting from the atomic relaxation.
[35] The slight differences are probably due to the use of a different XC energy functional at the DFT level (GGA in our case and LDA in the work of Fleszar and Hanke).